High-Pressure Syntheses of Novel Binary Nitrogen Compounds of Main Group Elements**

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In memoriam Holger Behrends (1975-2000)

The application of high-pressure methods in the search for novel materials usually requires additional effort compared to syntheses at ambient pressure. Depending on the desired p/T conditions different methods may be used. Special techniques and experimental apparatus such as shock waves, diamond anvil cells, and multianvil presses, which have been applied mainly by earth scientists and physicists in the past, are increasingly being applied by synthetic chemists and material scientists. A series of fascinating discoveries have been made recently as is demonstrated by three examples of binary nitrogen compounds: 1) Diazenides, compounds with N_2^{2-} ions, were obtained as single-phase products and structurally characterized for the first time. 2) At 11 GPa and 1800 K a phosphorus(v) nitride was prepared, which contains tetragonal PN₅ pyramids as a novel structural motif. 3) Macroscopic amounts of spinel silicon nitride were synthesized by shock-wave techniques, which allows the comprehensive characterization and possibly the implementation of this new hard material.

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- [**] The author thanks R. Riedel (TU Darmstadt) and F.F. Lange (University of California, St. Barbara) for their generous support, as well as A. Zerr (TU Darmstadt), S. Buhre (Universität Frankfurt), U. Hornemann (Freiburg), B. Poe (Universität Bayreuth), and P. McNutt (Rockland Research Inc.) for the useful information concerning costs and applications of the different high-pressure techniques. Furthermore, the author thanks U. Schwarz, G. Auffermann and R. Kniep (MPI für Chemische Physik fester Stoffe, Dresden), W. Schnick (Ludwig-Maximilians-Universität München), A. Simon (MPI für Festkörperforschung, Stuttgart), and P. Kroll (RWTH Aachen) for comments on the manuscript and providing unpublished results. Financial support by the Alexander von Humboldt Stiftung (Feodor Lynen scholarship), the Deutsche Forschungsgemeinschaft (DFG, Bonn), and the Fonds der Chemischen Industrie (FCI, Frankfurt) is gratefully acknowledged.

1. Introduction

Temperature, reaction time, and concentration or reactant ratios are without question the most widely used parameters in preparative chemistry. Another very crucial parameter for the successful performance and optimization of chemical reactions is the pressure, which is less frequently used. The major reason for this is certainly the additional experimental effort required for applying high-pressure methods.^[1] The "standard high-pressure technique" in the field of synthetic chemistry is based on the application of autoclaves and highpressure bombs. In recent years the application of these techniques was further expanded, for example for reactions in supercritical media.^[2] Methods for the generation of extremely high pressures (p > 1 GPa) have been developed in the fields of earth sciences. Currently, these techniques are increasingly used in preparative inorganic chemistry and material science, which is reflected by several recently published review articles.[3]

Table 1 shows a comparison of the presently most frequently used high-pressure techniques. Each of these methods represents a compromise with respect to maximum pressure, temperature, sample size, costs, and experimental effort. Autoclaves and high-pressure bombs are easy to operate, cheap, and allow the production of large samples; however, the p/T range is rather limited. "Large" presses such as belt or multianvil equipment (Figure 1) are much more expensive and more difficult to operate;^[1c,d] however, they allow pressures and temperatures up to 25 GPa and 3000 K, respectively, and produce amounts of sample that permits convenient characterization with the common analytical techniques. Even higher pressures and temperatures are achievable with laser-heated diamond anvil cells (Figure 2).^[1e,f] Owing to the transparency of the diamond anvils they offer the opportunity of in situ examinations, for example by vibrational spectroscopy or X-ray diffraction. Disadvantageous are the small sample sizes ($\ll 1 \text{ mg}$). Consequently, energy-dispersive X-ray analysis (EDX) and electron energy loss spectroscopy (EELS) measurements in transmission electron microscopes are usually necessary to estimate the elemental composition of the products.

Shock-wave syntheses methods^[1g,h] can be subdivided in three classes: 1) "flyer-plate" or "impact" methods, in which a projectile (flyer) is shot onto an especially prepared (mixture)

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Table 1. Overview of high-pressure techniques and apparatus currently used in preparative chemistry.

	Autoclaves/Pressure bombs	"Piston-cylinder-presses"	"Large" presses ^[a]	Shock syntheses ^[b]	Diamond anvil cells
p _{max} [GPa]	1.5	5	25 (40 ^[c])	150 ^[d]	500
$T_{\rm max}$ [K]	1200 (2000 ^[e])	2000	3000	5000	7000
sample size	1 kg (10 g) ^[f]	<10 g ^[f,g]	0.001 – 1 g ^[g]	0.1-500 g ^[h]	$\ll 1 \text{ mg}$
prime costs [€]	500-25000	2500-75000	50000-400000	> 25 000[i]	75000-150000 ^[j]
cost of operation	low	low-moderate	high	moderate-very high	low-moderate

[a] For example, multianvil presses. [b] For example, flyer plate or detonative syntheses. [c] Using diamond-based materials for the anvils. [d] By "impact" experiments much higher pressures may be generated, but usually it is very difficult to recover the sample. [e] Using IHPV ("internally heated pressure vessels"). [f] Depending on the pressure range different types of devices are used. [g] Sample sizes strongly depend on the desired pressure range. [h] For very large samples (>10 g), costs increase disproportionately, so that it is better to perform several experiments one after the other. [i] Strongly dependent on sample size and applied technique. [j] Including CO₂ or YAG laser for heating as well as equipment for in situ temperature and in situ pressure determination (Rubin-fluorescens).



Figure 1. Schematic sketch of a diamond anvil cell.



Figure 2. Schematic sketch of a two-stage cubic octahedral multianvil press.

of starting materials, 2) shock compressions or compactions, which usually are based on an explosive charge, which is in direct contact with the starting materials to be densified, producing a pressure wave, and 3) detonative syntheses, in which a precursor-explosive-mixture is ignited in a suitable vessel. All three techniques offer the opportunity to produce larger amounts of products. The disadvantages of shock-wave methods are obvious: The desired p/T condition can only be perpetuated for a short time (ca. 1 µs). Other frequently applied equipment, such as Bridgman or belt presses, was often modified and adapted to special requirements. Furthermore, there are other rarely used approaches such as the volume decrease upon solidification of melts. For all static high-pressure techniques the maximum pressure decreases with increasing temperature (strength decline) and increases with smaller sample sizes.

Similar to the field of oxides more and more research over the last few years has been concerned with ternary, quaternary and multinary nitrogen compounds.^[4] That the application of high pressure provides access to new interesting binary phases today is evidenced in the following by examples of novel nitrogen compounds of the main group elements.

2. High-Pressure Synthesis and Structural Characterization of Diazenides

Ionic oxygen compounds such as oxides (O²⁻), ozonides (O_3^{-}) , peroxides (O_2^{2-}) , and hyper- or superoxides (O_2^{-}) are well known;^[5] a comparison with nitride compounds raises the question, whether there are, in addition to nitride and azide ions, other anionic species-for example "pernitride" ions (N_2^{2-}) —that are stable in the condensed phase. There are numerous metal complexes known, in which the N2 ligands may be considered as N2²⁻ or N2⁴⁻ ions.^[6] This is particularly true for complexes of the early transition metals, which contain μ - η^1 : η^1 -bridging N₂ molecules. Furthermore, N₂²⁻ ions generated by ion implantation in CdS were detected by ESR spectroscopy.^[7] The high-pressure synthesis of BaN₂ was reported in 1953.^[8] Saltlike pernitrides of the alkaline earth metals containing the N24- ion have also been mentioned in literature.^[9] The existence of these compounds was supported by elemental analyses, IR spectra, as well as by the detection of hydrazine in the hydrolysis products.^[9] However, a structure determination of a compound with N_2^{x-} ions has not been accomplished yet.

Moreover, the number of binary nitrogen compounds of alkali metals and alkaline earth metals is still relatively small:^[10] For the Sr/N system, Sr_3N_2 , Sr_8N_5 , and SrN have been reported,^[10a] but only the subnitride $Sr_2N^{[10b,c]}$ and the azide $Sr(N_3)_2^{[10d]}$ have been structurally characterized. Against this background, the high-pressure synthesis and structural characterization of "diazenides" SrN and SrN_2 , that is compounds with N_2^{2-} ions, by Kniep et al.^[11] is particularly remarkable (Scheme 1). Phase-pure SrN was synthesized starting from Sr_2N at 920 K, 0.04 GPa, and a reaction time of 72 h. Increasing the pressure to 0.55 GPa under otherwise similar conditions furnished SrN_2 . The synthesis was carried out in a special autoclave, in which the initial N_2 pressure was 0.02 and 0.275 GPa, respectively.

$$\begin{array}{c} \begin{array}{c} +1.5 & -3 \\ 2 & \mathrm{Sr}_2 \left[\mathrm{N} \right] \xrightarrow{+ \mathrm{N}_2, \mathrm{a}} & +2 & -3 & -1 \\ \hline & & -\mathrm{N}_2, \mathrm{b} \end{array} \xrightarrow{+ 2 & \mathrm{N}_2 \left[\mathrm{N}_2 \right] \left[\mathrm{N}_2 \right] & \xrightarrow{+ 2 & \mathrm{N}_2, \mathrm{c}} & 4 & \mathrm{Sr} \left[\mathrm{N}_2 \right] \\ \rho = 3.5 \mathrm{g} \mathrm{cm}^{-3} & \rho = 3.8 \mathrm{g} \mathrm{cm}^{-3} & \rho = 4.2 \mathrm{g} \mathrm{cm}^{-3} \\ \hline & & -3 \mathrm{N}_2, \mathrm{d} \end{array}$$

Scheme 1. Synthesis and decomposition of the strontium diazenides SrN and SrN₂: a) 920 K, 0.04 GPa, 3 d; b) 573-673 K, 100 kPa; c) 920 K, 0.55 GPa, 3 d; d) 618-673 K, 100 kPa.

1433-7851/02/4101-0078 \$ 17.50+.50/0 Angew. Chem. Int. Ed. 2002, 41, No. 1

The structures of SrN and SrN₂, which are metastable at 20°C and 100 kPa, were determined by X-ray and neutron diffraction using powders. SrN₂ crystallizes isotypically to the corresponding alkaline earth acetylides in a tetragonal distorted NaCl structure (Figure 3, left). SrN is described as a nitride-diazenide: $(Sr^{2+})_4[N^{3-}]_2[N_2^{2-}]$. The structure can be derived from Sr₂N (inverse CdCl₂ type) by adding N₂²⁻ ions into the octahedral sites between the Sr₂N layers. The N-N distance within the diazenide ions (ca. 122.5 pm) corresponds well with that in μ - η^1 : η^1 -N₂ complexes, which may be considered to contain M-(N₂²⁻)-M units.^[6] The N-N distances in organic cis- (124.5 pm) and trans-dialkyl diazo compounds (122.2 pm) as well as the O-O distance in the oxygen molecule (120.7 pm) are also in the same region.^[12] Ab initio calculations for singlet- N_2^{2-} ions gave a value of 123.8 pm, which is also very close to the results found for the solid-state structures.^[13]

Simon and Kniep et al. have also reported the synthesis of barium pernitride, BaN2,^[14] which was obtained from the elements at 0.56 GPa and 920 K or by careful thermal decomposition of dry barium azide $Ba(N_3)_2$ at 100-300 kPa and 470-530 K. The crystal structure of BaN2 was again determined by powder X-ray and neutron diffraction. In contrast to the structure of SrN2 which contains N2 dumbbells oriented along the c axis and which crystallizes in the space group I4/mmm, the diffraction data for BaN₂ was more successfully refined in the space group C2/c. Here the octaherally coordinated N2 units are arranged perpendicular to the c axis and alternating by 90°. The N-N distance is almost identical to that for SrN₂. Interestingly, measurements of the magnetic susceptibility indicate that the substance is metallic, which, according to simulations of the band structure, is attributed to Ba(5d)-N(2p) interactions.

3. Synthesis of γ -P₃N₅ in a Multianvil Press: A Compound with Tetragonal PN₅ Pyramids

In contrast to the diazenides P/N phases should be interpreted as covalent inorganic polymers. They are similar to B/N and Si/N compounds in that they form tetrahedral EN_4 units, which can be linked to give oligomeric and polymeric

networks.^[15] Unlike for boron nitrides (*h*- and *c*-BN) and silicon nitrides (α - and β -Si₃N₄), the structure determination of the phosphorus nitrides PN and P₃N₅ turned out to be more difficult.^[16] Thus the structure of α -P₃N₅ was first resolved in 1997 by using powder diffraction with synchrotron radiation, after a synthesis route to phase-pure samples had been developed.^[17] This solid crystallizes in a novel structure type consisting of a three-dimensional network of corner- and edge-sharing PN₄ tetrahedra. Two fifths of the N atoms are threefold and three fifths are twofold coordinated with P atoms. The crystal structures of amorphous PN and β -P₃N₅ (according to high-resolution transmission electron microscopy (HRTEM) studies an ordered stacking variant of α -P₃N₅) are not known in detail.^[15–17]

Recently, a novel high-pressure phase of phosphorus(v) nitride γ -P₃N₅ was synthesized at 11 GPa and T=1800 K using the multianvil technique (Figure 1; Scheme 2).^[18] The

$$\begin{array}{cccc}
a & P_3N_5 & \xrightarrow{a} & \gamma P_3N_5 \\
\rho &= 2.77 \text{ g cm}^{-3} & \rho &= 3.65 \text{ g cm}^{-5}
\end{array}$$

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Scheme 2. Synthesis of γ -P₃N₅ in a multianvil press: a) 1773 K, 11 GPa, 5 min.

compound consists of one third PN₄ tetrahedra and two thirds tetragonal PN₅ pyramids. In contrast to the *a*-phase a larger fraction of the N atoms (i.e. four fifths) are threefold coordinated with phosphorus and only one fifth is bonded to two P atoms. Thus, starting from a ${}^{3}_{\infty}$ [P^[4]N^[2]N^[3]] network a ${}^{3}_{\infty}$ [P^[4]P^[5]N^[2]N^[3]] structure was formed (Figure 3, center). The three-coordinate N atoms are sp²-hybridized in both phases (sum of bond angles ca. 360°), while the P-N^[2]-P angle of 142 – 171° in *a*-P₃N₅ decreases to 111° in γ -P₃N₅.

The most remarkable feature of the P_3N_5 high-pressure phase is the novel PN_x coordination geometry. Although distorted trigonal PN_5 bipyramids have been found in molecular compounds, square or tetragonal PN_5 pyramids were not known previously.^[18] Combined with the increase of the mean coordination numbers, an increase of the average bond length is expected due to the pressure – distance rule; besides, a 32% decrease of the volume is observed. A similar increase of the coordination number combined with a



Figure 3. Left: Crystal structure of SrN_2 ;^[11] the layers at the top and bottom edges represent (N_2) $Sr_{6/3}$ octahedra, while a ball-and-stick-illustration is used in between; red: Sr^{2+} , green N_2^{2-} . Center: Polyhedral representation of the three-dimensional network structure of γ - P_3N_5 ; ^[18] red: PN_5 pyramids, blue: PN_4 tetrahedra; right: Si_3N_4 spinel structure^[24a, b]; green: N; yellow: octahedrally coordinated Si; orange: SiN_4 tetrahedra.

corresponding volume decrease was found for the recently discovered high-pressure phase of Si₃N₄ (see Section 4). From an initial ${}_{\infty}^{3}$ [Si₃^[4]N₄^[3]] network a ${}_{\infty}^{3}$ [Si₂^[6]Si^[4]N₄^[4]] structure was formed at 12 GPa and 1600 K.

The high-pressure phases of Si_3N_4 and P_3N_5 are potential materials for engineering ceramics due to their temperature resistance, hardness, and chemical inertness. This statement is most likely also valid for related ternary and multinary nitride high-pressure phases such as spinel-SiAlONs^[19] as well as for nitridophosphates^[20] and possibly also for nitridosilicates and nitridoborates.

4. Spinel-Si₃N₄ Synthesized by Shock-Wave Methods

Until 1999 there were six binary EN_x phases of the heavy elements of the fourth main group known: α - and β -Si₃N₄, α and β -Ge₃N₄ as well as α - and β -Pb(N₃)₂. Additionally, certain subnitrides have been reported, which, however, are frequently amorphous and structurally not well characterized, but nevertheless are partly used as functional materials.^[21] In addition, there are numerous molecular CN_x compounds known.^[22] However, the existence of a hard material with the molecular formula C₃N₄, that is a carbon(iv) nitride, has not been confirmed despite several reports about its successful synthesis.^[23]

In 1999 three novel spinel nitride phases of the elements Si, Ge, and Sn were discovered almost simultaneously.^[24] Whereas c-Si₃N₄ (γ -Si₃N₄, see Figure 3, right)^[24a,b] as well as c-Ge₃N₄ $(\gamma$ -Ge₃N₄)^[24c,d] were synthesized in diamond anvil cells and multianvil presses at $p \ge 12$ GPa, the preparation of Sn₃N₄^[24e]—the first binary Sn/N-compound—was accomplished at ambient pressure. Owing to the importance^[25] of β silicon nitride as a ceramic material for cutting tools, machine and engine components, as well as electronic parts, the discovery of another Si₃N₄ modification attracted much interest.^[26] The increase in density (4.0 g cm⁻³) by about 25% compared to the formerly known modifications (3.2 gcm⁻³) and a bulk modulus of about 310 GPa might indicate an application of the cubic Si₃N₄ phase as a hard material. Syntheses on a larger scale than possible in diamond-anvil cells or multianvil presses are required for further characterization of the materials properties and possibly the fabrication of c-Si₃N₄-based components for practical utilization. Currently, the equipment used in industry for the commercial preparation of diamond and c-BN is limited to a maximum pressure of 6 GPa,^[27] which is not sufficient for the synthesis of c-Si₃N₄. An alternative to static high-pressure techniques are dynamic methods, which have been used successfully to produce the ultra-hard materials c-BN and diamond.^[1g,h]

Already shortly after the discovery of spinel silicon nitride Sekine et al. obtained c-Si₃N₄ in high yields by shock-wave methods (Scheme 3).^[28] Submicrometer powders of pure β -Si₃N₄ or a α -/ β -Si₃N₄ mixture were blended with copper. Targets wrapped in copper or platinum hulls were cannonaded with steel or platinum projectiles.^[28a] By varying the projectile velocity between 1.5 and 2.1 km s⁻¹ pressures of 12 to 115 GPa and temperatures above 3000 K were generated

Scheme 3. Synthesis of γ -Si₃N₄ from β -Si₃N₄ by shock-wave methods: a) 3000 K, 50 GPa, 1 µs. The formation from α -Si₃N₄ under the same conditions is not successful.

for about one microsecond. The products were characterized after the removal of the copper with HNO₃ by X-ray structure analysis, transmission electron microscopy, and EDX. A maximum *c*-Si₃N₄ yield of 80%(!) was obtained at 50 GPa and 2400 K (decomposition of the silicon nitride was observed at still higher pressures and temperatures). The particle size of the *c*-Si₃N₄ lay between 10 and 50 nm. Interestingly, α -Si₃N₄ does not transform into the spinel phase under the same conditions. This indicates a diffusionless (martensitic) transformation from β - to *c*-Si₃N₄.

In another report from the research group of Sekine^[28b] sintered dense β -Si₃N₄ discs (1 × 1.2 × 0.3 cm) were cannonaded with polycarbonate, steel, and tungsten projectiles. The minimum pressure of 36 GPa required to produce c-Si₃N₄ in these experiments is much higher than that for the static experiments, which is attributed to the low temperature of 460 K. A complete transformation into the high-pressure phase was not observed even at the highest pressure of 150 GPa generated in the experiments, but a 100 % conversion was extrapolated for $p_{\text{max}} = 180$ GPa. However, according to recent calculations for this pressure range, a further highpressure modification should be formed, which is similar to the CaTi₂O₄ structure, and contains SiN₆ prisms as well as SiN₆ octahedra.^[29] Recently published diamond anvil cell studies indicate that at 34 GPa/RT a further new Si₃N₄ phase already forms: δ -Si₃N₄.^[30] However, the structure of this compound has not been solved so far.

A thorough design and analysis of the impact experiments allowed a determination of the pressure/density function (Hugoniot) for Si₃N₄. Using these data and the Birch–Murnaghan equation of state a value of 300 ± 10 GPa was estimated for the bulk compression modulus. This result corresponds well with the results of theoretical predictions^[24a] as well as recent static measurements of the compressibility of c-Si₃N₄ in a diamond anvil cell.^[31]

5. Conclusions

It is clear since the development of the ammonia synthesis and the preparation of *c*-BN that the application of high pressure is very useful for obtaining EN_x compounds and materials. Interesting new solids are still being synthesized by high-pressure techniques today: Kniep et al. succeeded in confirming that besides the three formerly known homopolyatomic nitrogen species N₂, N₃⁻, and N₅⁺ also N₂²⁻ ions are stable in the solid phase.^[11] Schnick et al. demonstrated that in addition to PN₄ tetrahedra also tetragonal PN₅ pyramids occur as structural motifs in P–N-compounds.^[18] Sekine et al. have shown that dynamic techniques may be used to produce and characterize compounds, which are available only in very small amounts with static techniques.^[28]

Another group of EN_x high-pressure phases of main group elements are the nitrides AlN, GaN, and InN with rock salt structure.^[32] These phases are noteworthy given the enormous interest in the corresponding Wurzite phases for semiconductor research, and it is surprising that very little has been reported on these materials to date.^[32]

Of course, the fascinating results that have been obtained recently by high-pressure techniques are not limited to *binary* nitrogen compounds of main group elements. For example, the discovery of a diamondlike BC_2N phase, which—shortly after the structural analysis of a graphite BC_2N modification obtained at 3-5 GPa^[33]—was prepared in a multianvil press at 18 GPa and 2200 K, is a milestone in hard materials research.^[34] This first cubic B/C/N phase now represents the second hardest substance after diamond.

Even for "simple" compounds such as elemental nitrogen, elemental carbon in the form of fullerenes, or carbon dioxide, highly interesting results were obtained upon compression in diamond anvil cells. At a pressure of 140 GPa polymeric N₂ was formed. The product is stable at ambient pressure up to a temperature of 100 K.^[35] In a similar manner CO₂ was transformed at >35 GPa and 1800 K into a solid with a tridymite-like SiO₂ structure, which is detectable at room temperature up to a pressure of 1 GPa.^[36] Depending on the *p*/*T* conditions, fullerenes are polymerized to give one-, two-, and three-dimenional (sometimes crystalline) networks,^[37] which are characterized, for example, by very interesting magnetic properties.^[38]

The examples presented here show how useful the application of high pressure is for preparative chemistry, and that the additional costs and effort can be worthwhile. Further propagation and application of high-pressure techniques will certainly produce interesting and useful materials in the future.

Received: June 20, 2001 [M1520]

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