



17<sup>th</sup> International Conference on Atomic Layer Deposition  
Denver, Colorado, USA  
2017

# What theory can tell us about ALD mechanism

Simon D. Elliott & co-workers



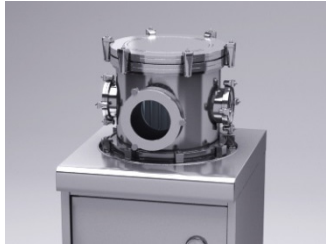
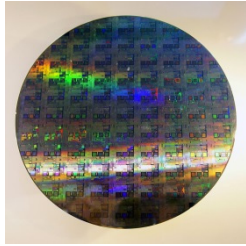
[simon.elliott@tyndall.ie](mailto:simon.elliott@tyndall.ie)

[www.tyndall.ie](http://www.tyndall.ie)

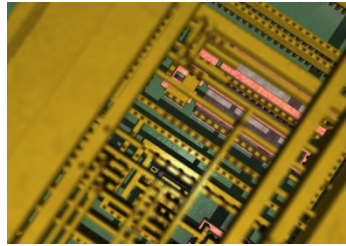
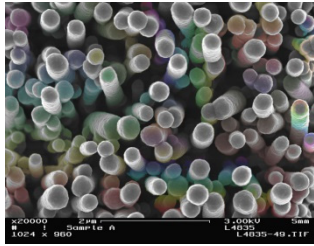


# The length scales of ALD

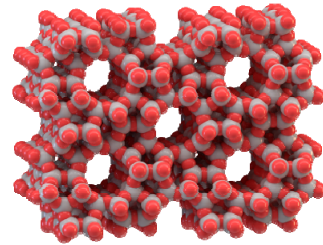
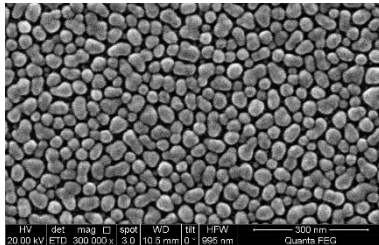
metre



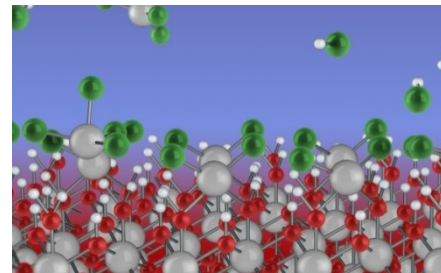
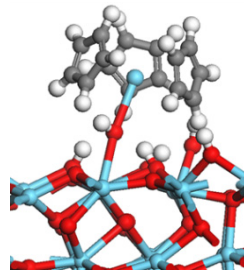
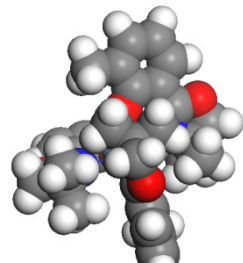
micro



nano



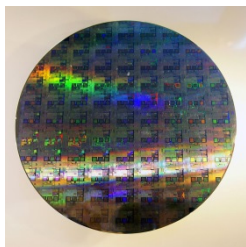
atomic





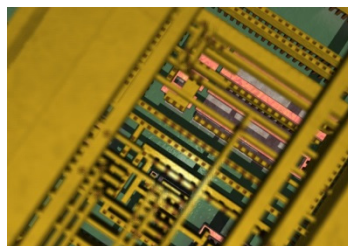
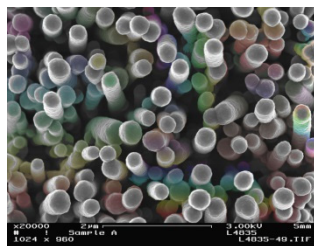
# The length scales of ALD modelling

metre



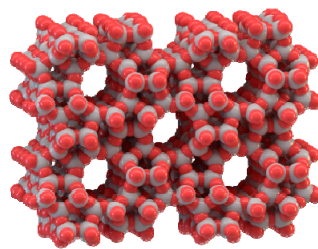
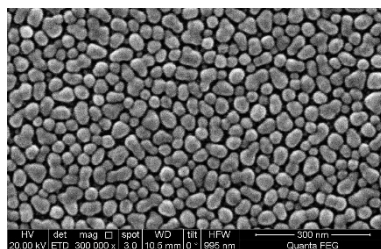
Computational fluid dynamics

micro



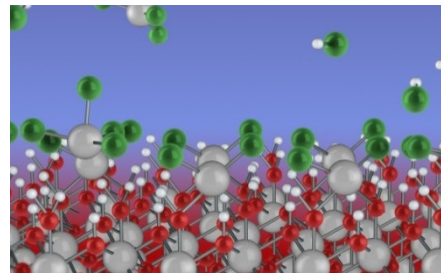
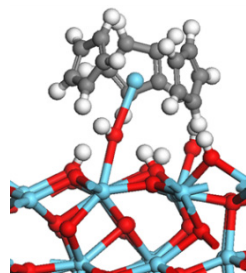
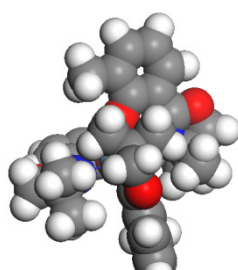
Coarse grained models

nano



Atomistic simulations

atomic

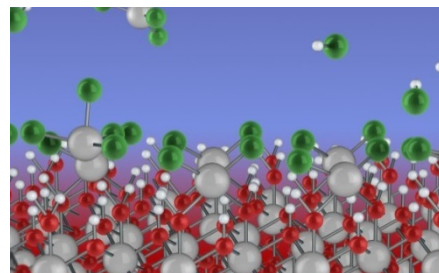
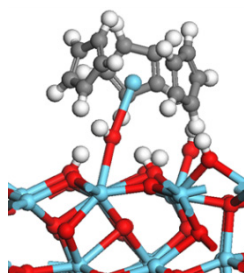
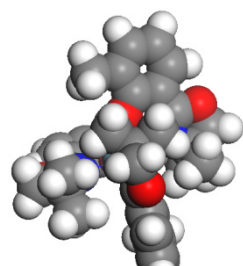


Density functional theory



# Atomic scale modelling for ALD

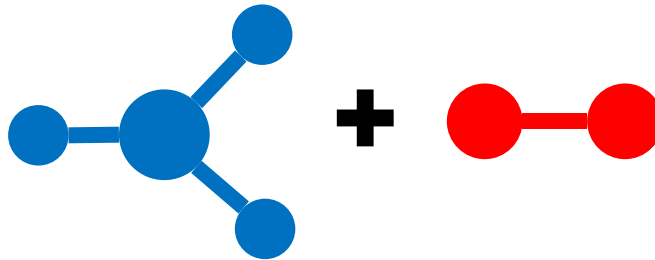
atomic



Density  
functional  
theory



# Precursor molecules

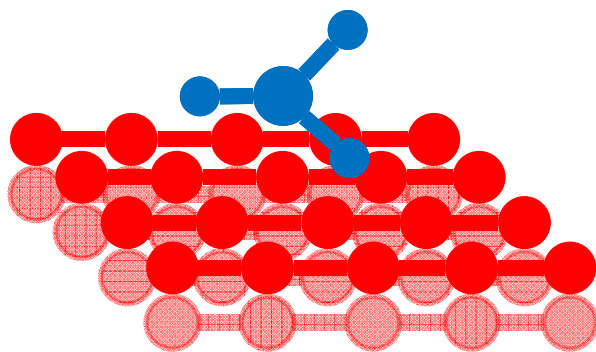


## **PRECURSOR PROPERTIES**

S. D. Elliott, *Semicond. Sci. Technol.* **27**, 074008 (2012);  
S. D. Elliott *et al.*, *Adv. Mater.* **28**, 5367 (2016).



# Adsorption onto simple surfaces



## SUBSTRATE MODELS

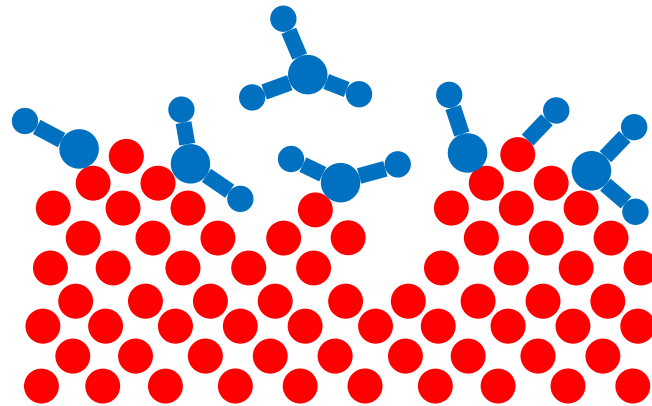
S. D. Elliott, *Semicond. Sci. Technol.* **27**, 074008 (2012);  
S. D. Elliott et al., *Adv. Mater.* **28**, 5367 (2016).





# Adsorbates on 3D-structured surfaces

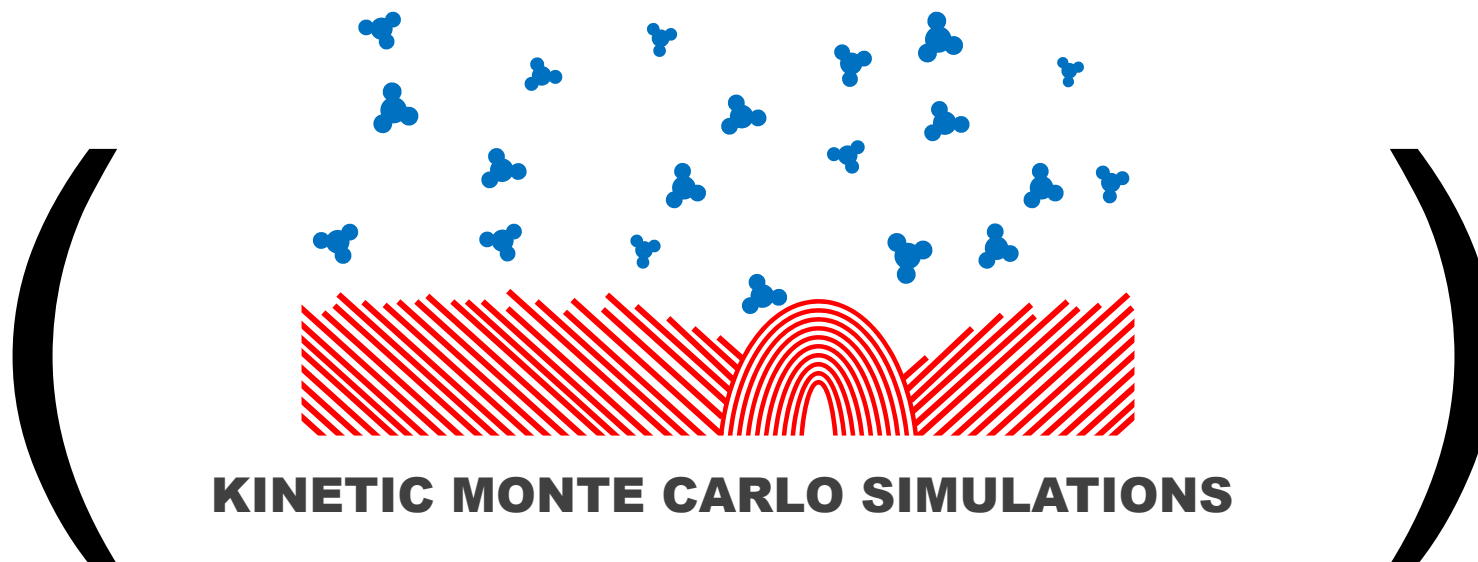
## MULTIPLE ADSORBATES



S. D. Elliott, *Semicond. Sci. Technol.* **27**, 074008 (2012);  
S. D. Elliott *et al.*, *Adv. Mater.* **28**, 5367 (2016).



# Gas flow and film morphology



**KINETIC MONTE CARLO SIMULATIONS**

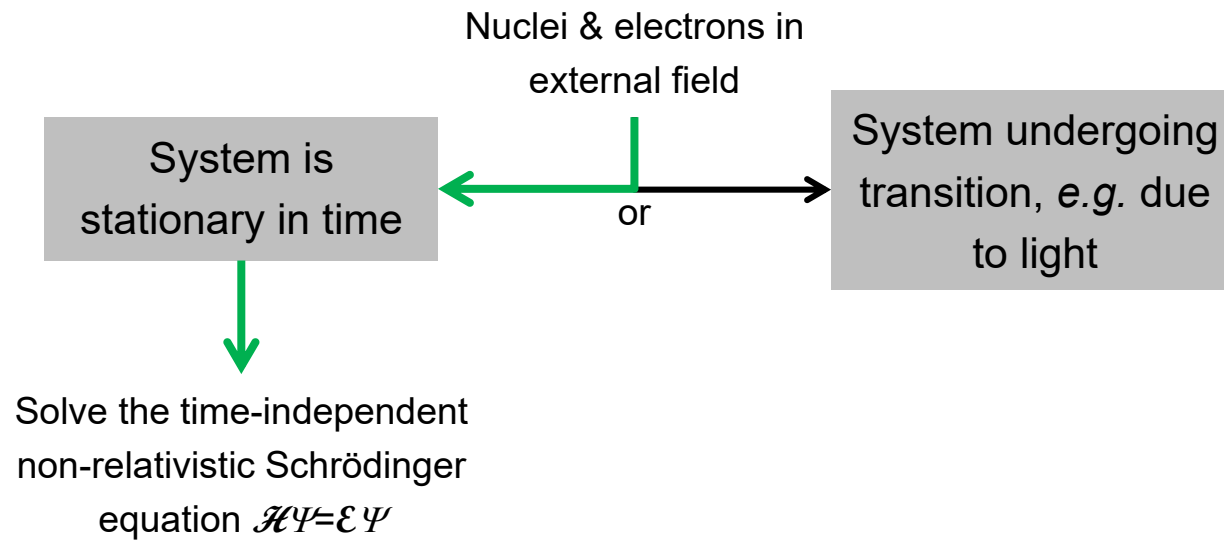
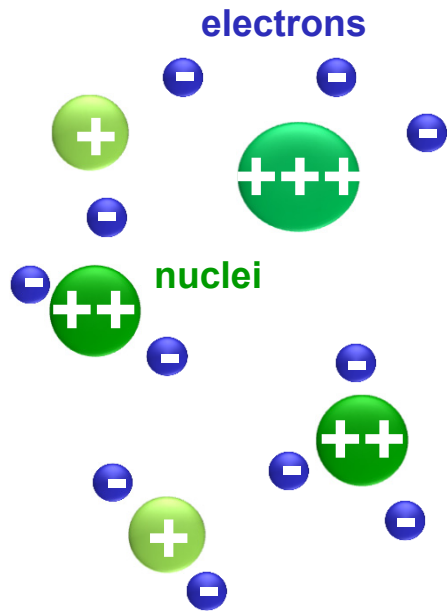
**You**  <sup>TM</sup> <https://www.youtube.com/watch?v=tR1AQicsLRY>

S. D. Elliott, *Semicond. Sci. Technol.* **27**, 074008 (2012);  
S. D. Elliott et al., *Adv. Mater.* **28**, 5367 (2016).



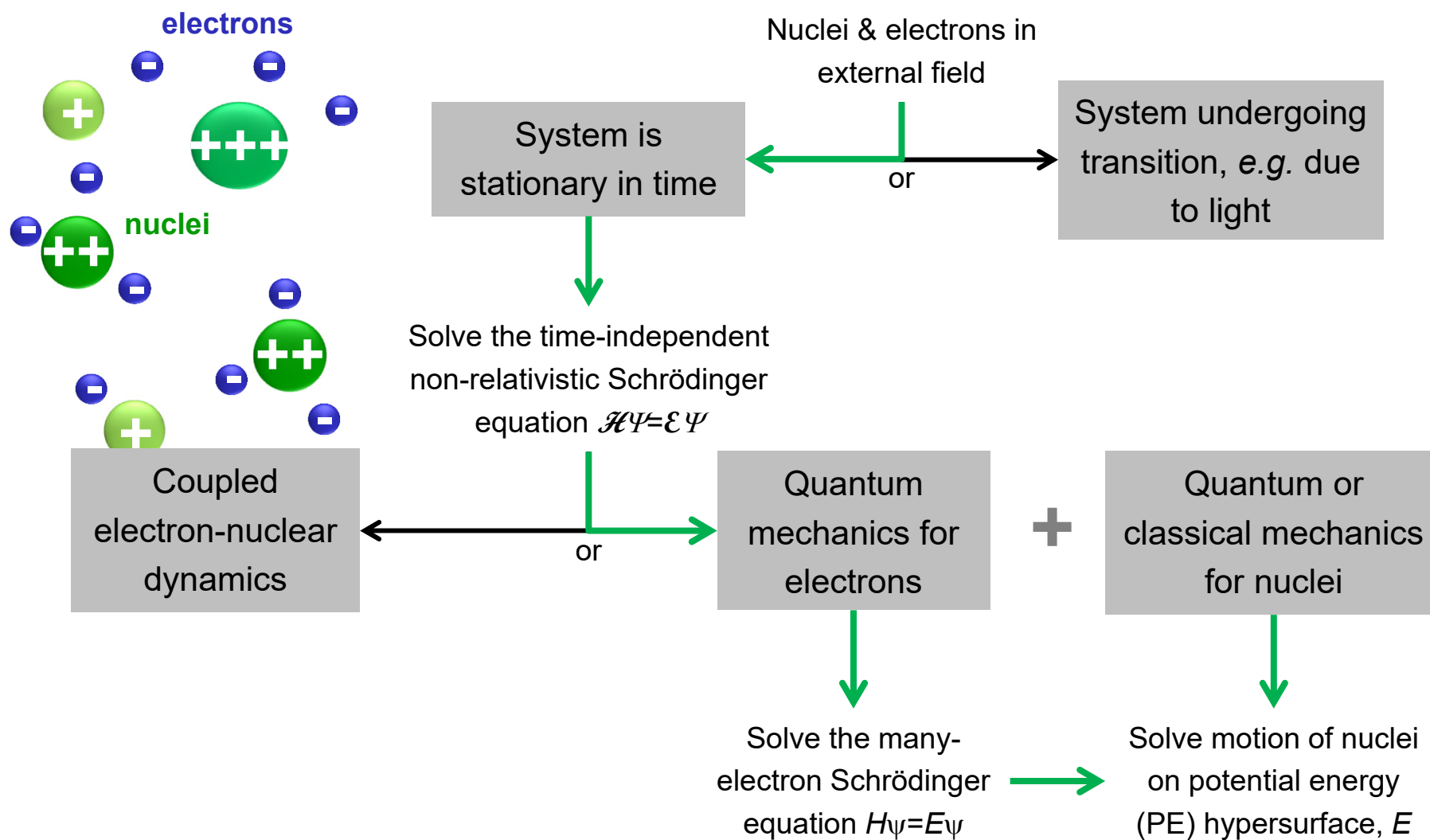


# Atomic-scale simulations



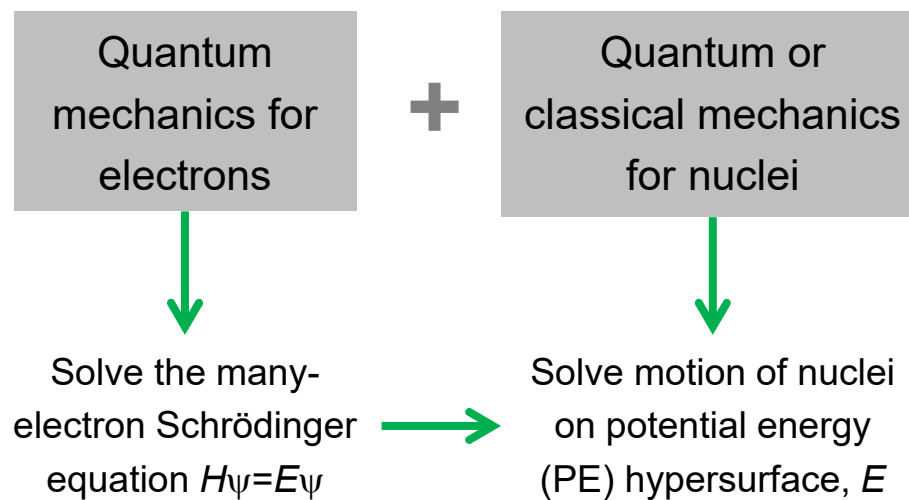
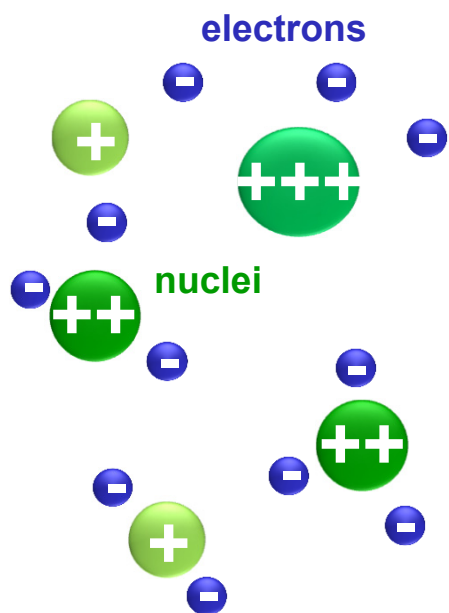


# Atomic-scale simulations



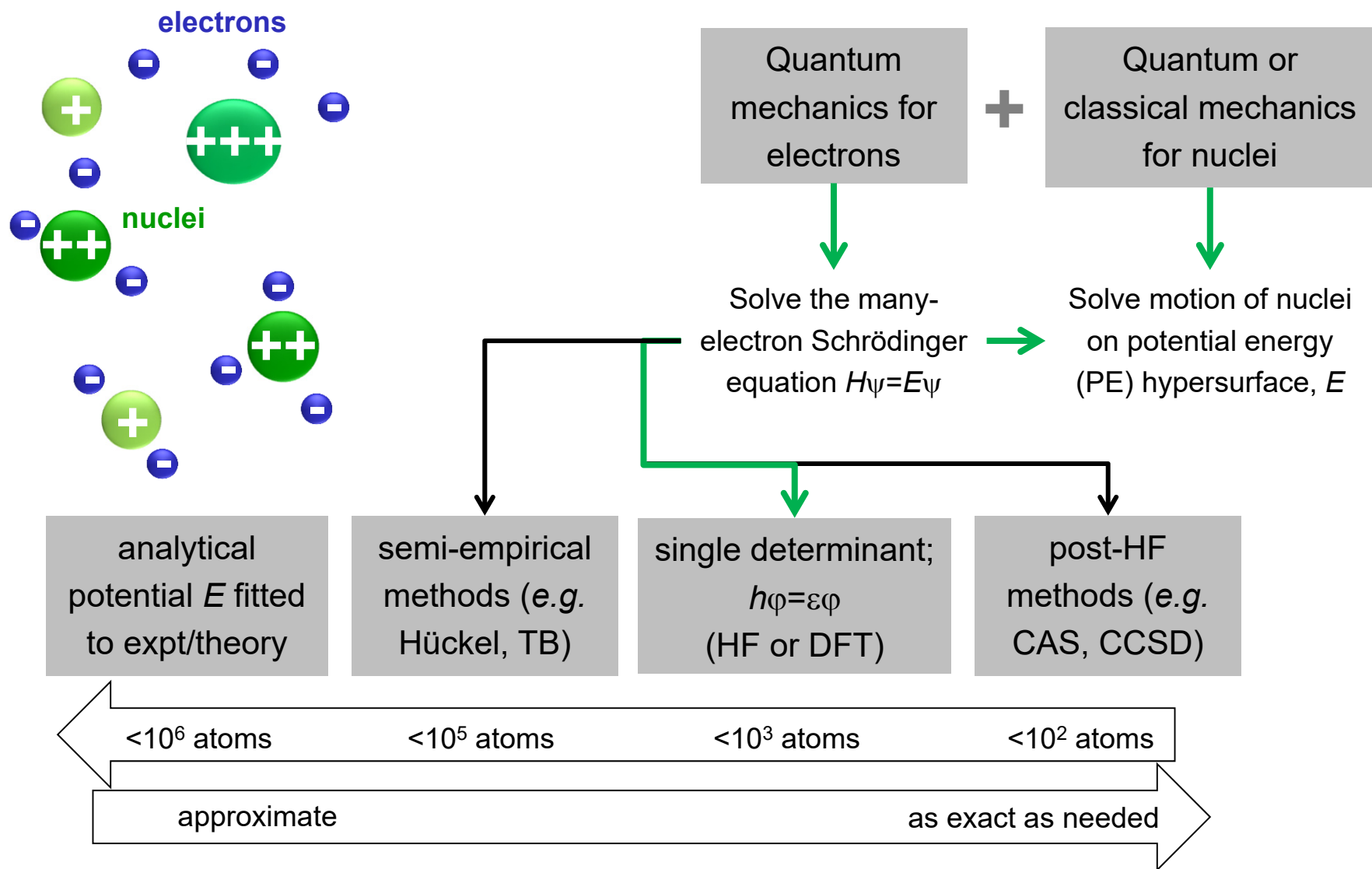


# Atomic-scale simulations





# Atomic-scale simulations





# Atomic-scale simulations

analytical  
potential  $E$  fitted  
to expt/theory

semi-empirical  
methods (e.g.  
Hückel, TB)

single determinant;  
 $h\phi = \epsilon\phi$   
(HF or DFT)

post-HF  
methods (e.g.  
CAS, CCSD)

Fixed potential energy function, varying only with interatomic geometry.



Analogy: always bring an umbrella,  
just in case it will rain.



# Atomic-scale simulations

analytical  
potential  $E$  fitted  
to expt/theory

semi-empirical  
methods (e.g.  
Hückel, TB)

single determinant;  
$$h\phi = \epsilon\phi$$
  
(HF or DFT)

post-HF  
methods (e.g.  
CAS, CCSD)

Wavefunction-based description of electrons, but solutions fitted to known data.



Analogy: bring umbrella if weather statistics for this day averaged over many years indicate high likelihood of rain.

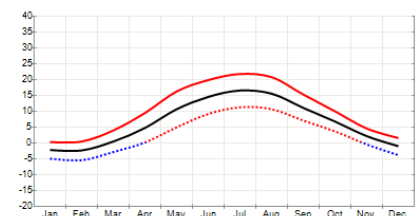
## Weather statistics for Linköping, Östergötland (Sweden)

Add to My places Forecast as PDF

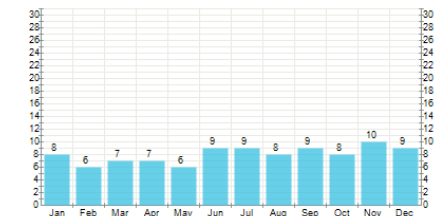
- Overview
  - Hour by hour
  - Long term
  - Statistics
  - Maps
- RELEVANT PLACES
- Linköping
  - Tórshavn
- To the main menu at the page bottom

Malmslätt weather station: 93 m.a.s.l., 5.0 km away from Linköping  
[Jönköping, 111.0 km fra Linköping](#)  
[Jönköping flygplats, 115.0 km fra Linköping](#)

Average temperature per month



Average days with precipitation per month





# Atomic-scale simulations

analytical  
potential  $E$  fitted  
to expt/theory

semi-empirical  
methods (e.g.  
Hückel, TB)

single determinant;  
 $h\phi = \epsilon\phi$   
(HF or DFT)

post-HF  
methods (e.g.  
CAS, CCSD)

Many-electron wavefunction is anti-symmetric product of many one-electron wavefunctions (“orbitals”).

## HARTREE-FOCK THEORY:

Solve one-electron problem in mean field due to other electrons

→ Missing dynamical correlation

Insist that each orbital is empty or doubly-occupied

→ Missing static correlation (e.g. when HOMO=LUMO)



Analogy: bring  
umbrella if rain is  
forecast for today.

Long term forecast for

## Linköping, Östergötland (Sweden)

Overview

Hour by hour

Long term

Statistics

Maps

RELEVANT PLACES

### Long term forecast

Monday 5 June 12-18	Tuesday 6 June 12-18	Wednesday 7 June 12-18	Thursday 8 June 14-20	Friday 9 June 14-20
15°	21°	13°	14°	14°
0 mm	0 mm	0 mm	4.7 mm	5.2 mm





# Atomic-scale simulations

analytical  
potential  $E$  fitted  
to expt/theory

semi-empirical  
methods (e.g.  
Hückel, TB)

single determinant;  
 $h\psi = \epsilon\psi$   
(HF or DFT)

post-HF  
methods (e.g.  
CAS, CCSD)

Many-electron wavefunction is anti-symmetric product of many one-electron wavefunctions (“orbitals”).

## DENSITY FUNCTIONAL THEORY:

Solve one-particle problem in mean field that mimics electron correlation  
→ Some dynamical correlation included

Insist that each orbital is empty or doubly-occupied

→ Missing static correlation (e.g. when HOMO=LUMO)

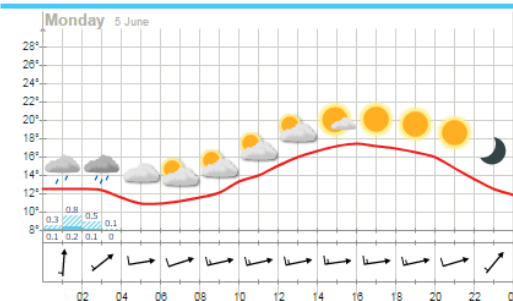


Analogy: bring umbrella if  
hour-by-hour weather  
forecast indicates high  
likelihood of rain now.

## Hourly forecast for Linköping, Östergötland (Sweden)

- Overview
- Hour by hour**
- Detailed
- Long term
- Statistics
- Maps
- RELEVANT PLACES
- Linköping
- Copenhagen

### Meteogram, next 48 hours





# Atomic-scale simulations

analytical  
potential  $E$  fitted  
to expt/theory

semi-empirical  
methods (e.g.  
Hückel, TB)

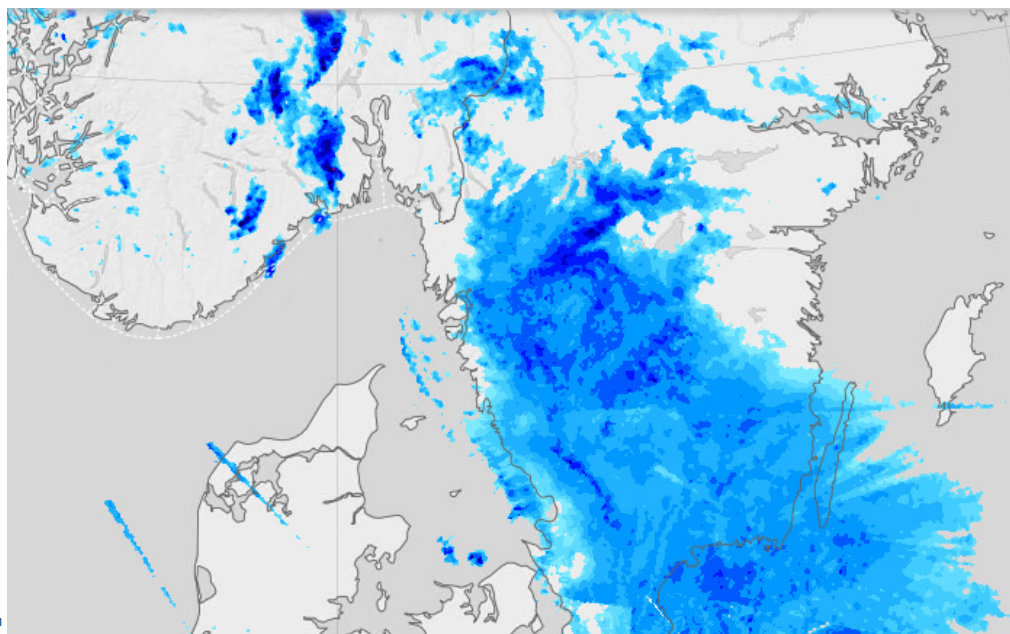
single determinant;  
 $h\varphi = \varepsilon\varphi$   
(HF or DFT)

post-HF  
methods (e.g.  
CAS, CCSD)

Many-electron wavefunction by mixing configurations and/or reference states from single-electron solutions.



Analogy: bring  
umbrella if  
radar shows  
that rain is  
approaching.



## SHARE

## RESEARCH ARTICLE



0



2

# Reproducibility in density functional theory calculations of solids

Kurt Lejaeghere<sup>1\*</sup>, Gustav Bihlmayer<sup>2</sup>, Torbjörn Björkman<sup>3,4</sup>, Peter Blaha<sup>5</sup>, Stefan Blügel<sup>2</sup>, Volker Blum<sup>6</sup>, Damien Calis...

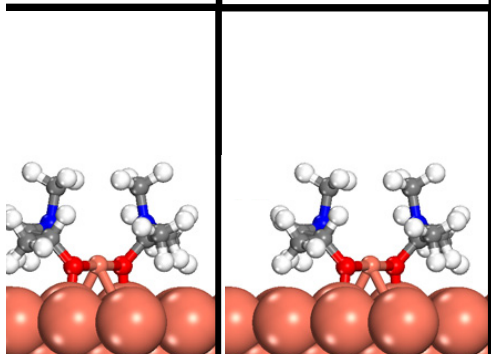
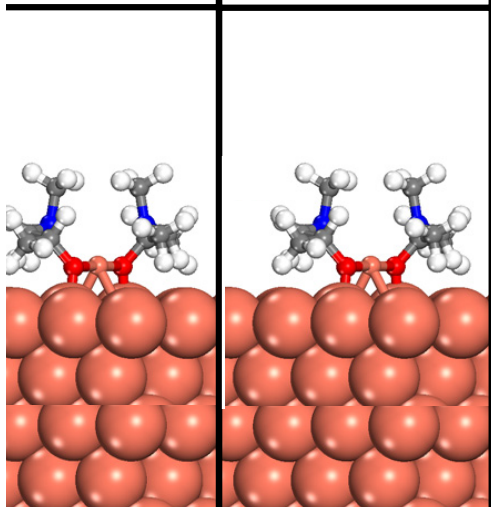
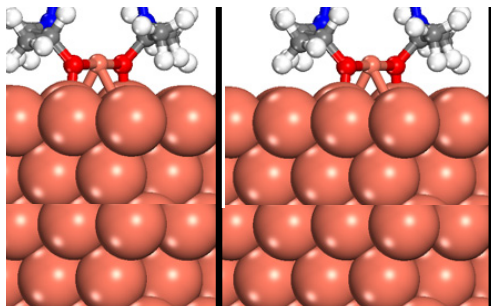
+ See all authors and affiliations

*Science* 25 Mar 2016:  
Vol. 351, Issue 6280, aad3000  
DOI: 10.1126/science.aad3000



Peer Reviewed  
← see details

Lejaeghere *et al.* compared the calculated values for the equation of states for 71 elemental crystals from 15 different widely used DFT codes employing 40 different potentials. Although there were variations in the calculated values, most recent codes and methods converged toward a single value, with errors comparable to those of experiment.

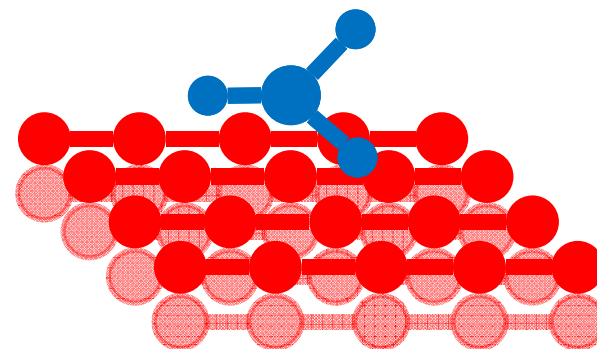


## Density functional theory for slabs

$10^3$  atoms per cell

Length scale  $10^{-9}$  m

Time scale  $10^{-12}$  s



Surface represented by infinitely-repeating 4 layer slab of (111)-oriented fcc-Cu separated by vacuum;

- VASP program;
- Functional due to Perdew, Burke & Ernzerhof with vdW-optB88 correction;
- Plane waves to 450 eV cutoff;
- PAW treatment of cores;
- k-point at  $\Gamma$  is adequate for large  $p(6 \times 6)$  cell with 18 Å vacuum;
- Self consistent steps to  $10^{-4}$  eV;
- Geometry optimised by conjugate-gradient;
- Activation energies  $E_a$  by nudged elastic band.

**VASP:** Kresse, G.; Hafner, J. *J. Phys.: Condens. Matter* 1994, 6, 8245.

**PBE:** Perdew, J. P.; Burke, K.; Ernzerhof, M. *Phys. Rev. Lett.* 1996, 77, 3865–3868.

**Slab:** Payne, M. C.; Teter, M. P.; Allan, D. C.; Arias, T. A.; Joannopoulos, J. D. *Rev. Mod. Phys.* 1992, 64, 1045–1097.

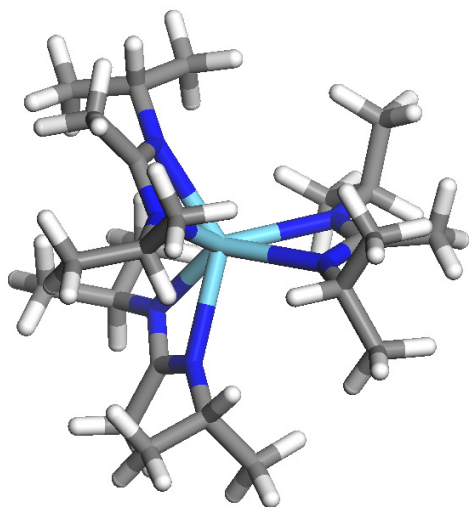
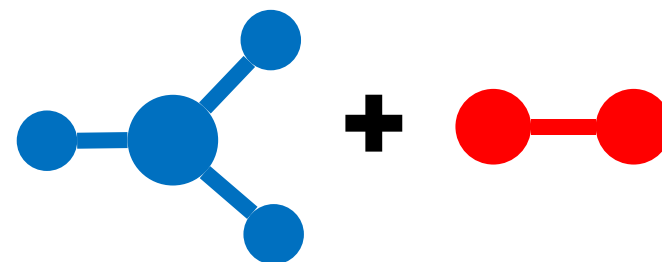


# Density functional theory for molecules

$10^3$  atoms in vacuum

Length scale  $10^{-9}$  m

Time scale  $10^{-12}$  s



- Isolated cluster of atoms in gas phase;
- TURBOMOLE program;
- Functional due to Becke & Perdew (1986) or Perdew, Burke & Ernzerhof (1996);
- Basis set of localised atomic orbitals: def2-TZVPP;
- Effective core potentials on heavy atoms;

**TURBOMOLE:** Ahlrichs, R.; Bär, M.; Häser, M.; Horn, H.; Kölmel, C. *Chem. Phys. Lett.* 1989, 162, 165.

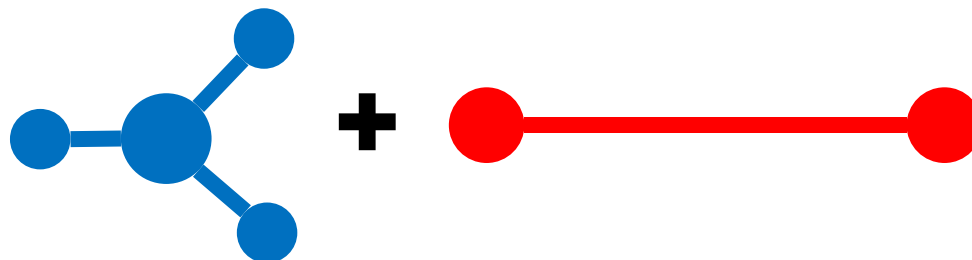
Becke, A. D. *Phys. Rev. A* 1988, 38, 3098; Perdew, J. P. *Phys. Rev. B* 1986, 33, 8822.

Perdew, J. P.; Burke, K.; Ernzerhof, M. *Phys. Rev. Lett.* 1996, 77, 3865–3868.

Eichkorn, K.; Weigend, F.; Treutler, O.; Ahlrichs, R. *Theor. Chem. Acc.* 1997, 97, 119.



H. Ablat



## Hybrid ALD - MLD

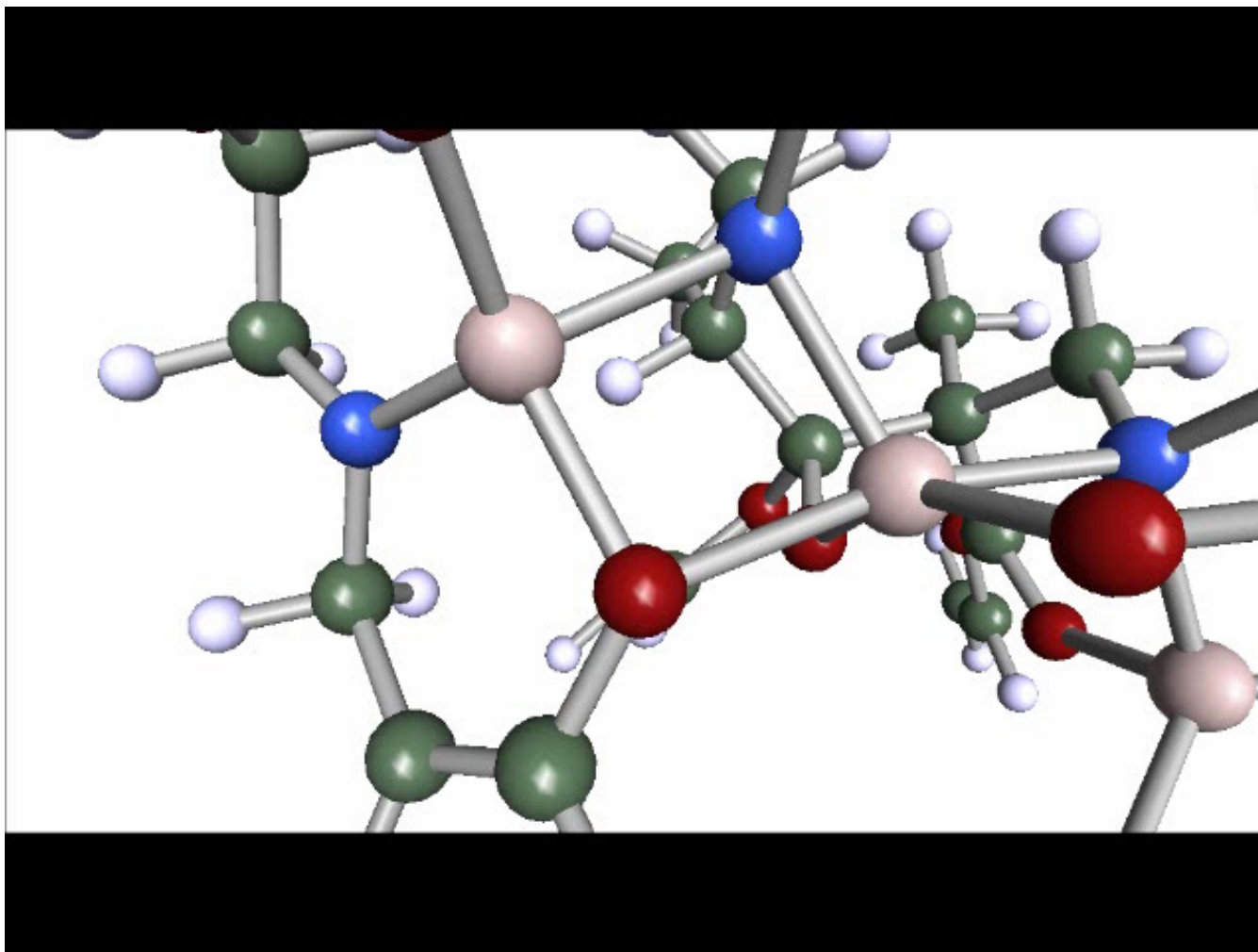
TMA + acrylate + ethanolamine







## Bonding in hybrid MLD/ALD films



O-Al-N stretch  
computed at  
 $511 \text{ cm}^{-1}$

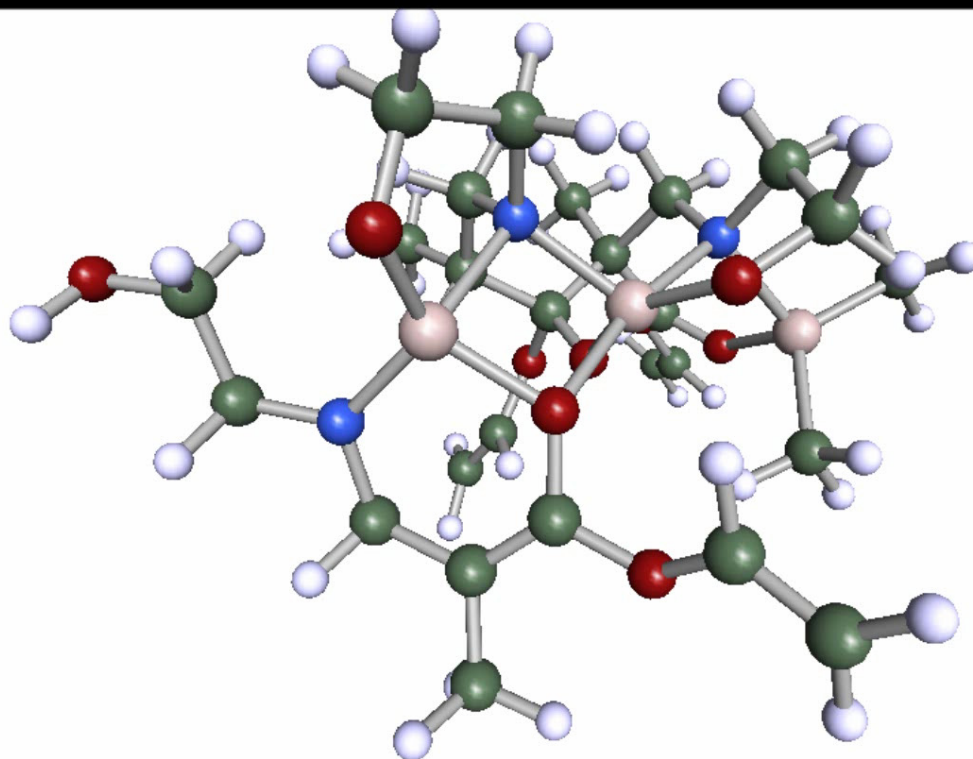
...measured  
with FTIR at  
 $514 \text{ cm}^{-1}$







## Bonding in hybrid MLD/ALD films



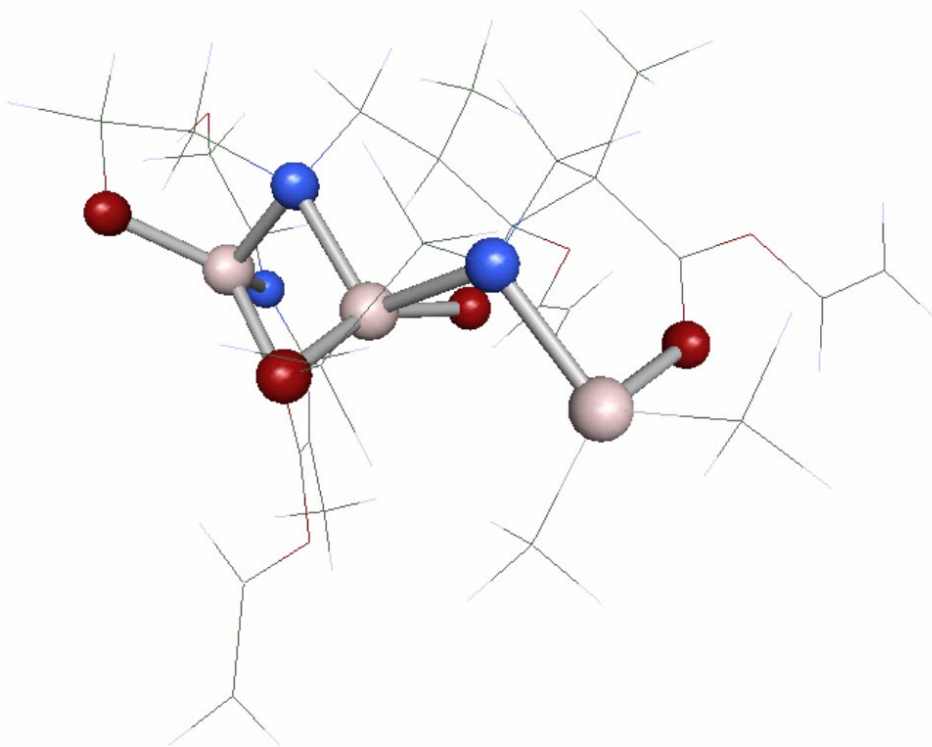
O-Al-N stretch  
computed at  
 $610\text{ cm}^{-1}$

... measured at  
 $610\text{ cm}^{-1}$  with  
FTIR





## Bonding in hybrid MLD/ALD films



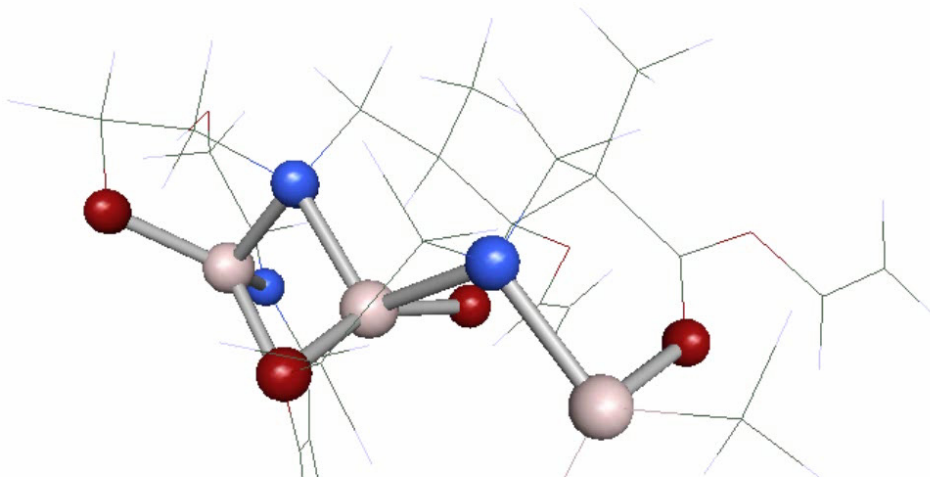
O-Al-N stretch  
computed at  
 $610\text{ cm}^{-1}$

... measured at  
 $610\text{ cm}^{-1}$  with  
FTIR





## Bonding in hybrid MLD/ALD films

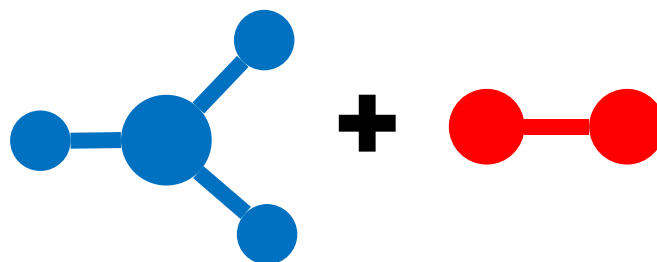


O-Al-N stretch  
computed at  
 $610\text{ cm}^{-1}$

... measured at  
 $610\text{ cm}^{-1}$  with  
FTIR

Little known about structure of exposed 'surface'  
during polymer MLD.

DFT can reveal how oligomers form.



## Combinatorial approach to precursor design

ALD of  $\text{Si}_3\text{N}_4$

Journal of

**THE KOREAN CERAMIC SOCIETY**



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BROWSE ARTICLES

CURRENT ISSUE

FOR AUTHORS AND REVIEWERS



J. Korean Ceram. Soc. > Volume 53(3); 2016 > Article

**Special Issue : Recent Advances in Computational Materials Science**

**Article**

Journal of the Korean Ceramic Society 2016; 53(3): 317-324.

Published online: May 31, 2016

DOI: <https://doi.org/10.4191/kcers.2016.53.3.317>

## **Quantum Mechanical Simulation for the Analysis, Optimization and Accelerated Development of Precursors and Processes for Atomic Layer Deposition (ALD)**

Thomas Jeffrey Lomax Mustard, Hyunwook Shaun Kwak<sup>\*</sup>, Alexander Goldberg, Jacob Gavartin<sup>\*\*</sup>, Tsuguo Morisato<sup>\*\*\*</sup>, Daisuke Yoshidome<sup>\*\*\*</sup>, Mathew David Halls<sup>†</sup>

Schrödinger Inc., San Diego, California 92121, United States

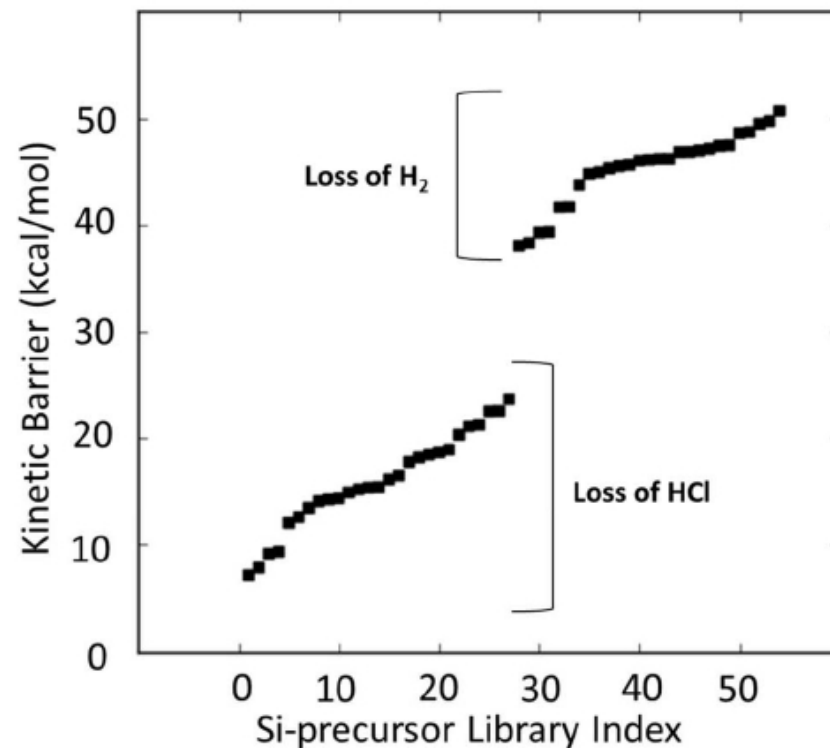
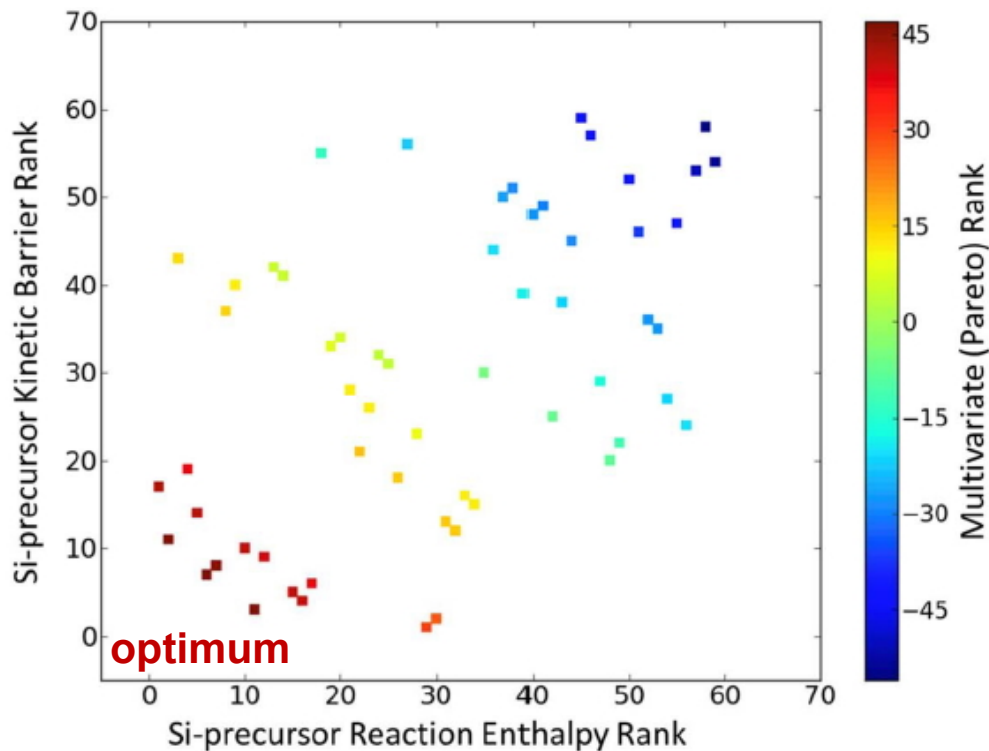
<sup>\*</sup>Schrödinger Inc., Cambridge, Massachusetts 02142, United States

<sup>\*\*</sup>Schrödinger Inc., Cambridge, Cambridgeshire CB1 2JD, United Kingdom

<sup>\*\*\*</sup>Schrödinger K.K., Chiyoda-ku, Tokyo 100-0005, Japan

<sup>†</sup>Corresponding author : Mathew David Halls, E-mail : [mat.halls@schrodinger.com](mailto:mat.halls@schrodinger.com)  
299-4532

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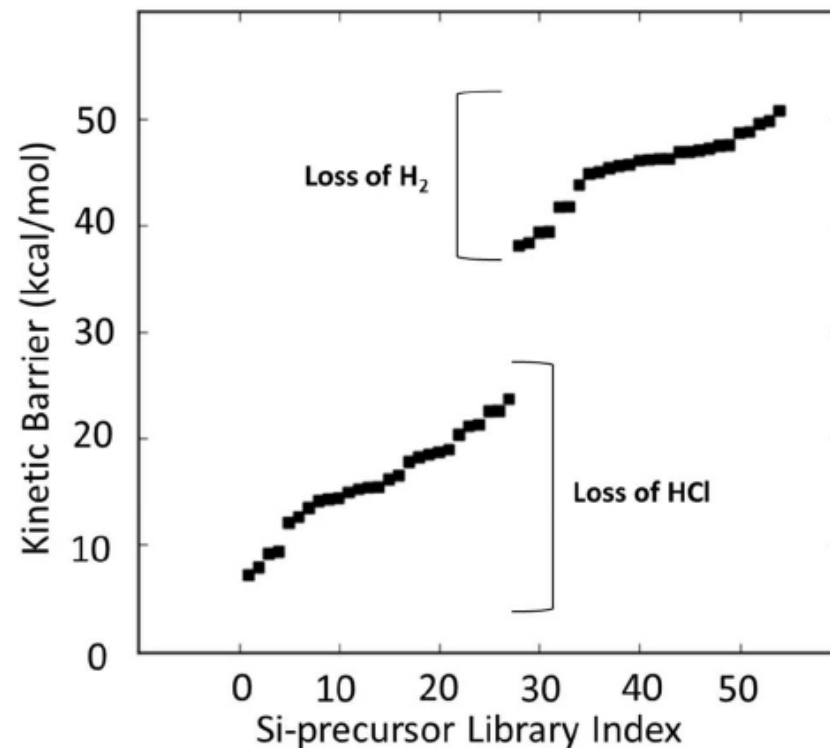
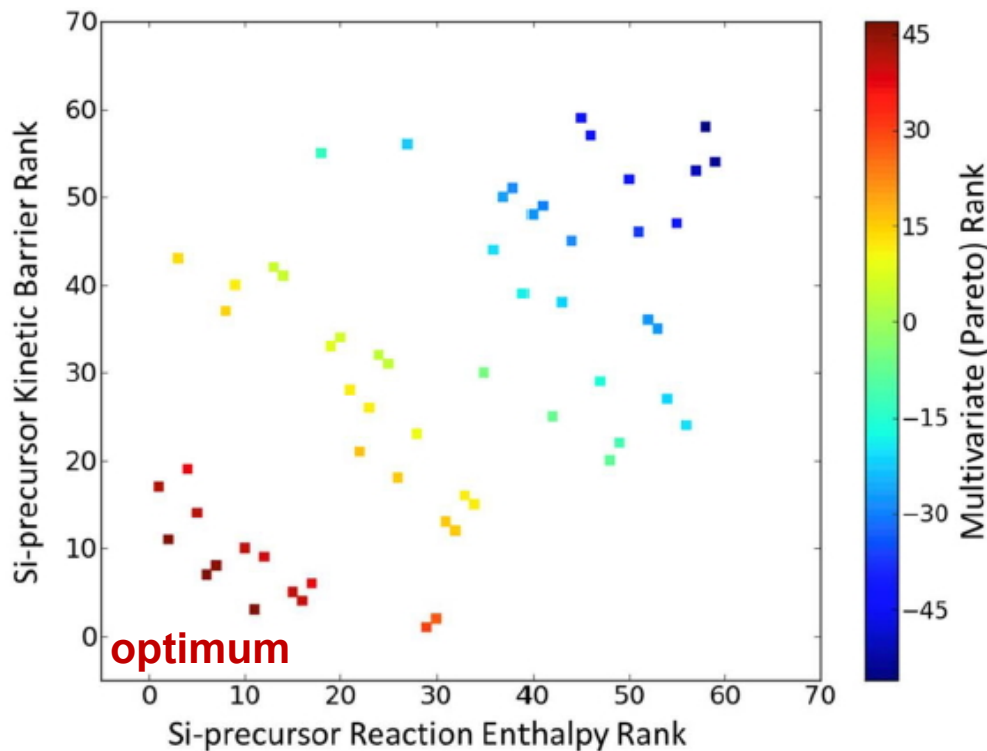
### Precursors screened:

Si(Cl)R<sub>2</sub> with R=H, Cl or CH<sub>3</sub>

Si<sub>2</sub>(Cl)R<sub>5</sub> with R=H or Cl

### Enthalpy and kinetic barrier evaluated for model reactions:

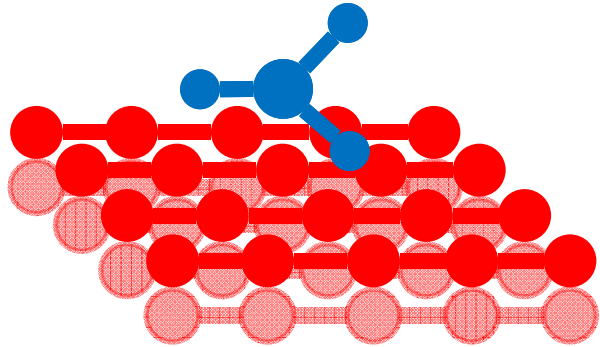




Knowledge about ALD mechanism allows metrics for precursor design to be defined.

Large numbers of precursors can be automatically generated and screened.





## Improved models of substrates and growing surfaces

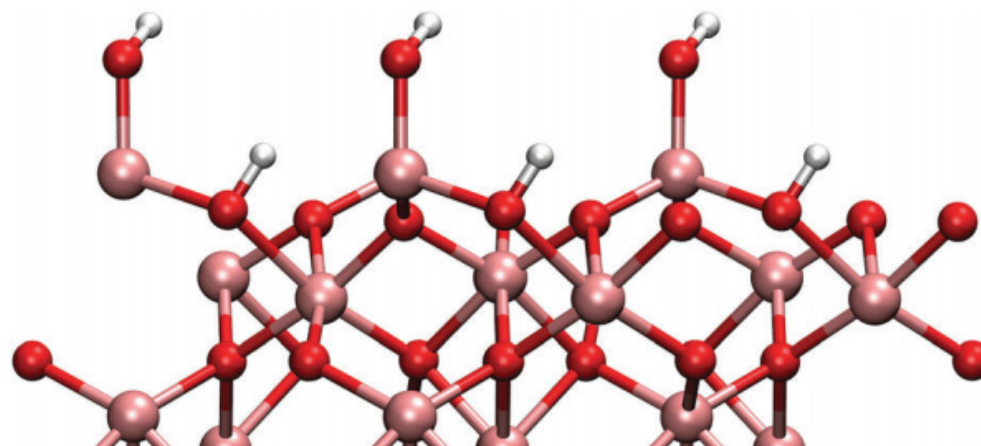
ALD of  $\text{Al}_2\text{O}_3$



Cite this: *Phys. Chem. Chem. Phys.*,  
2015, 17, 17322

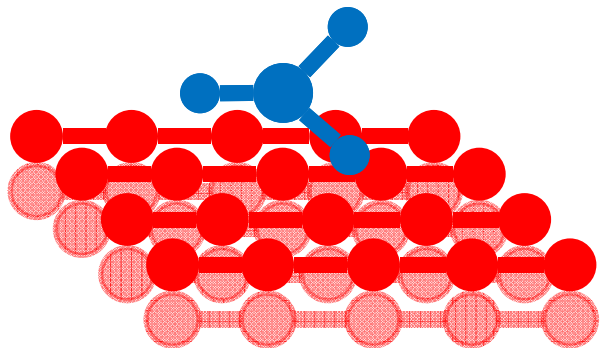
# First principles study of the atomic layer deposition of alumina by TMA–H<sub>2</sub>O-process

Timo Weckman\* and Kari Laasonen



**Fig. 1** Side-view of the hydroxylated surfaces formed from dissociatively adsorbed water. Two kinds of hydroxyl groups are present, the higher group formed from the water molecule and the lower one formed with the dissociated hydrogen and surface oxygen. A monolayer coverage is achieved when four water molecules adsorb onto a 2 × 2 cell which

pink = Al, red = O, white = H



C. Murray

## How adsorption energy affects exposure required for saturation

ALD of  $\text{Si}_3\text{N}_4$



## “Growth rate” in ALD?

### Individual reaction steps

- activation energies
- kinetics of individual steps
- kinetics of overall growth process.

### Rate at which growth process takes place

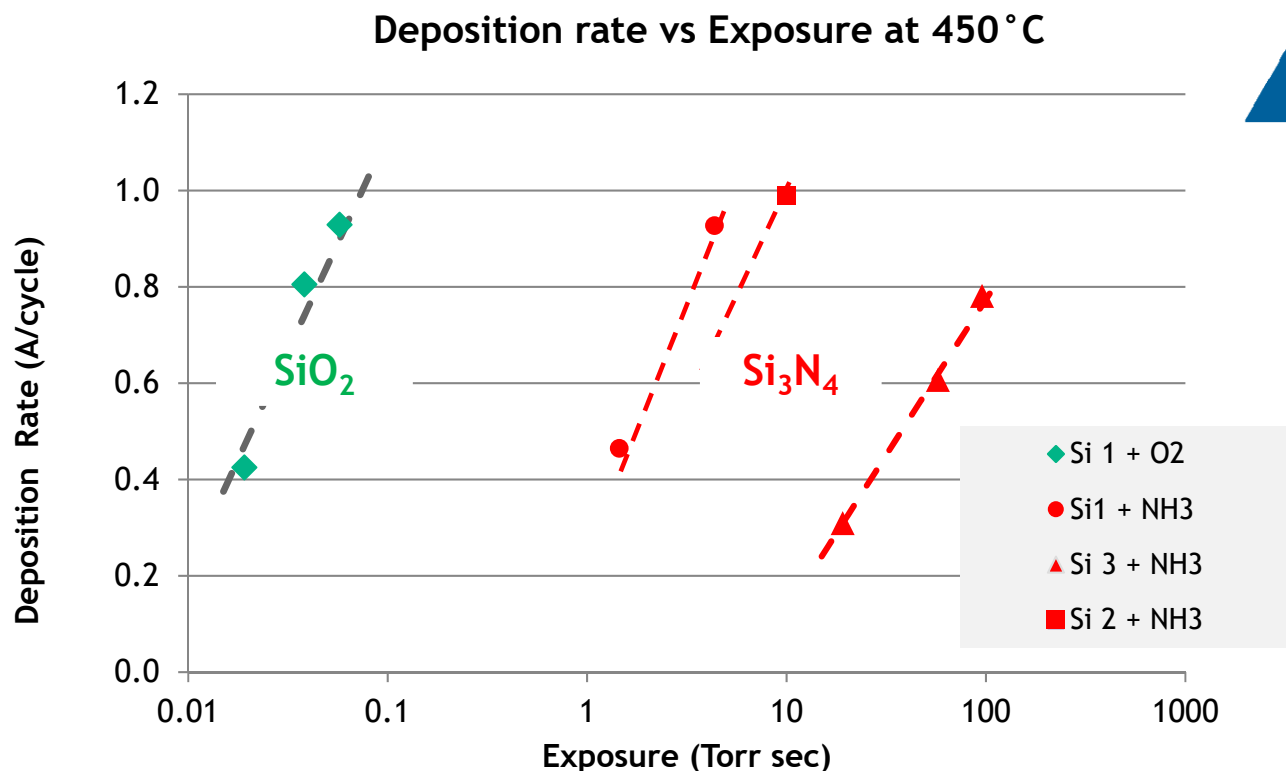
- exposure needed
- dependence of growth on process time
- throughput.

### Self-limiting surface chemistry

- saturation
- maximum thickness increment per cycle (GPC).



# Reactivity of ALD $\text{SiO}_2$ versus $\text{Si}_3\text{N}_4$

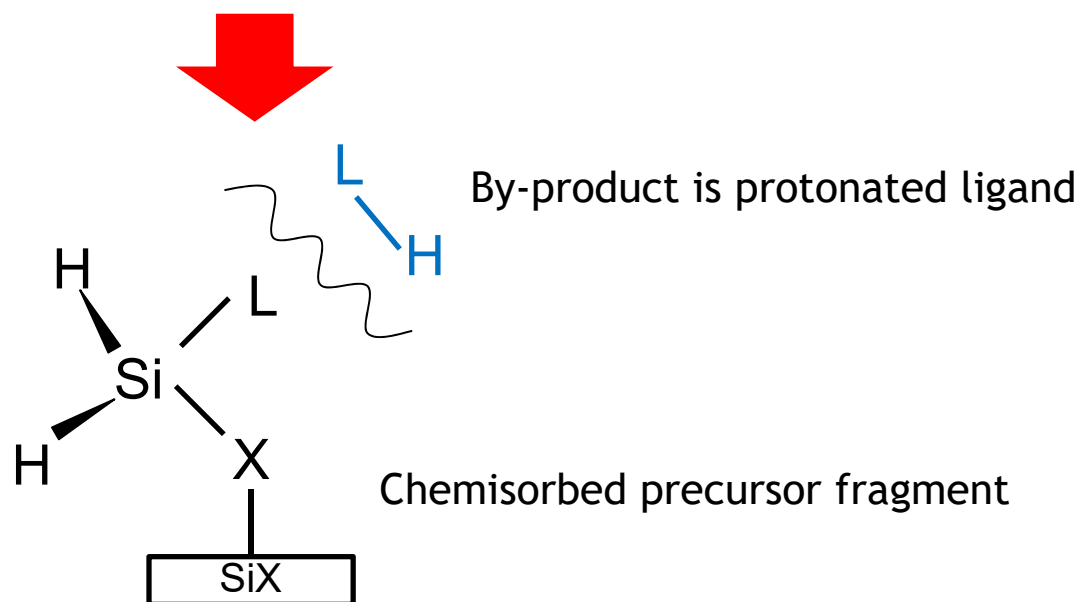
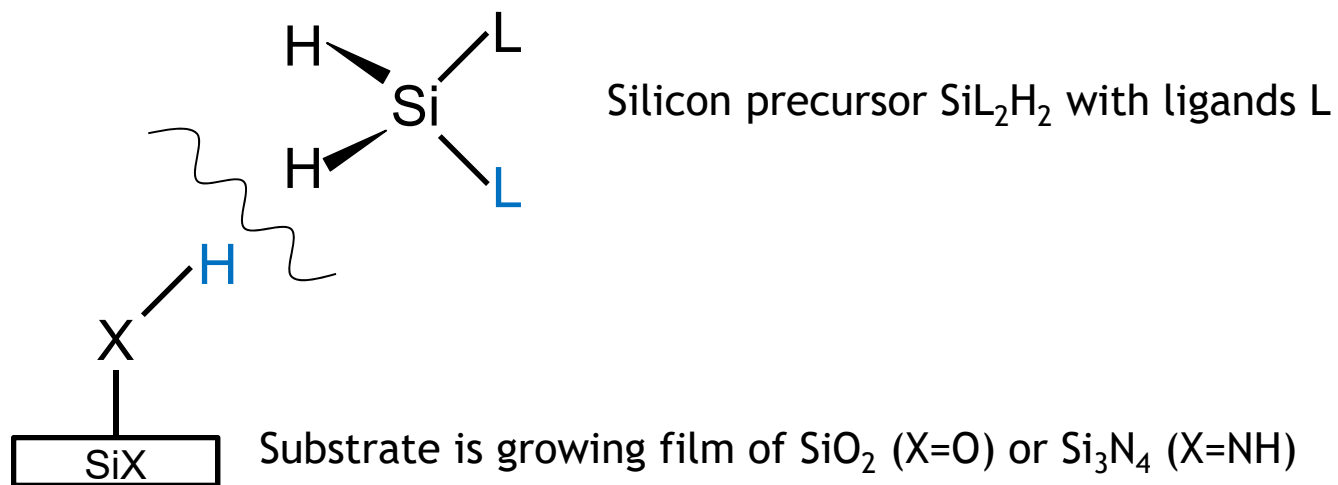


Require 100x greater exposure of Si precursor to achieve saturation for  $\text{Si}_3\text{N}_4$  relative to  $\text{SiO}_2$

**What Si precursor can solve this problem?**



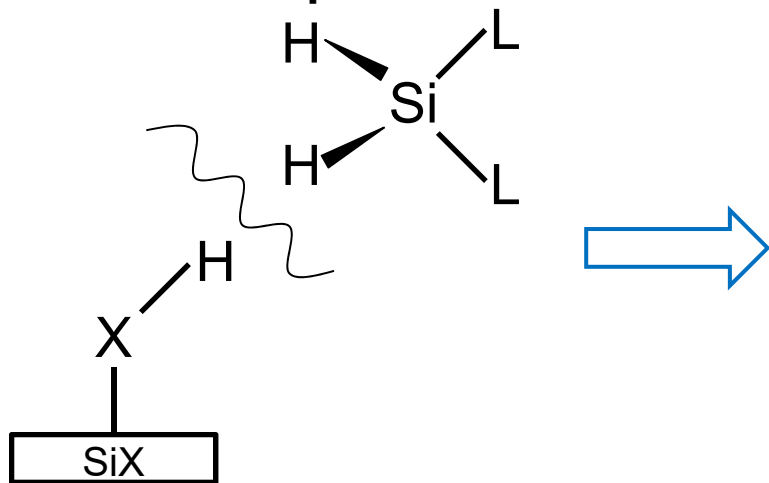
# Reactivity of ALD $\text{SiO}_2$ versus $\text{Si}_3\text{N}_4$



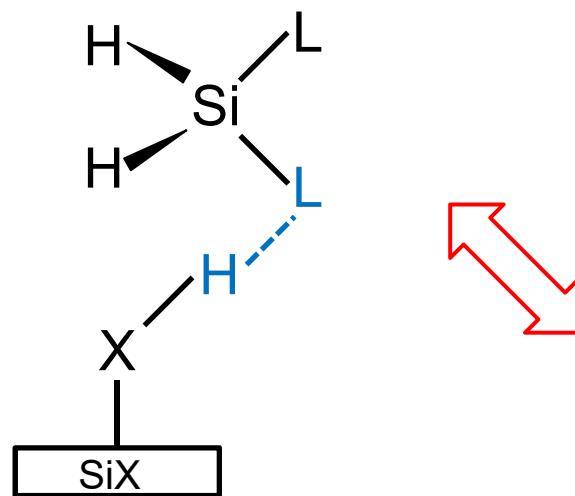


# Reactivity of ALD $\text{SiO}_2$ versus $\text{Si}_3\text{N}_4$

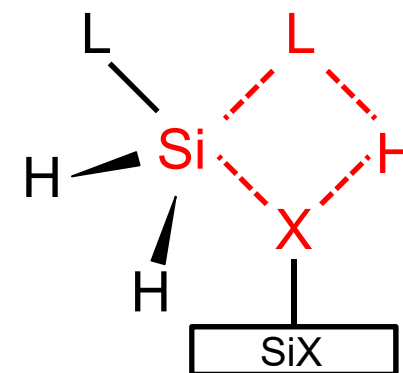
Unbound precursor



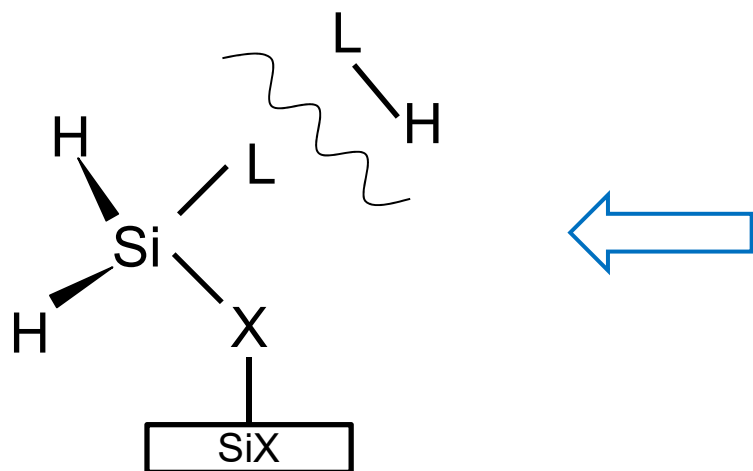
Bound precursor



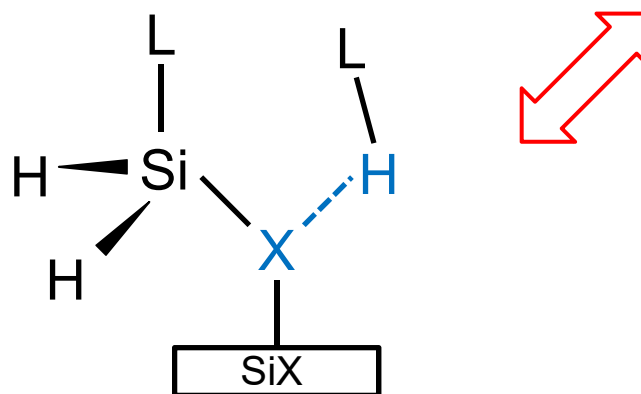
Transition State



Desorption of by-product



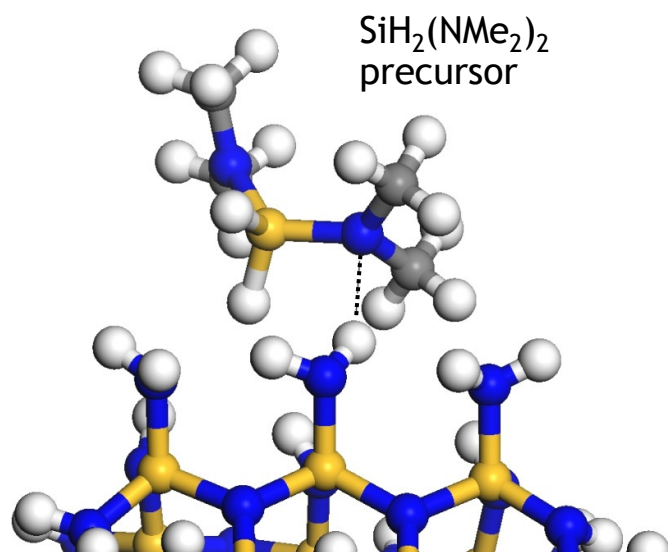
Bound by-product



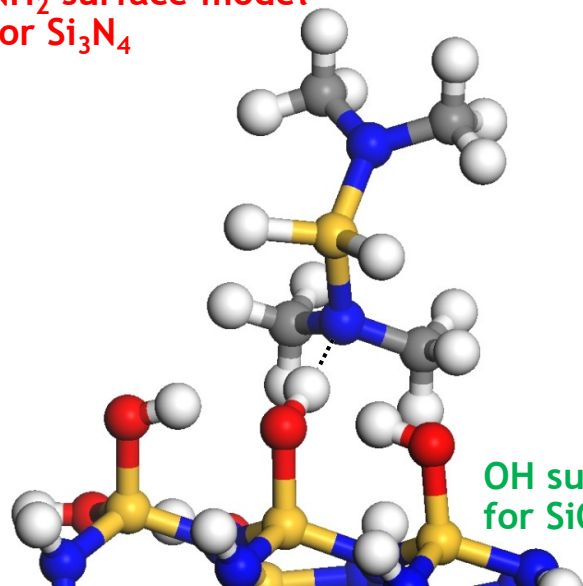




# Reactivity of ALD $\text{SiO}_2$ versus $\text{Si}_3\text{N}_4$

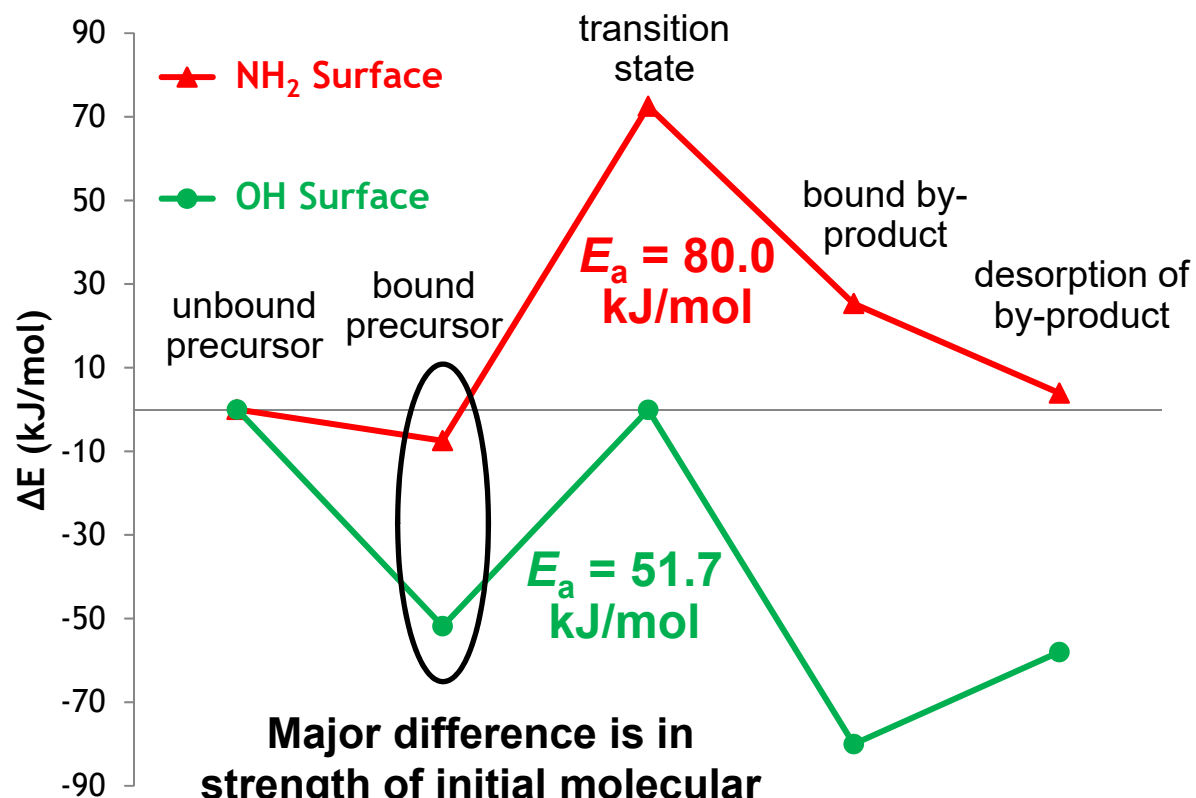


$\text{NH}_2$  surface model for  $\text{Si}_3\text{N}_4$



$\text{OH}$  surface model for  $\text{SiO}_2$

BP86/SV(P) at  $T = 0 \text{ K}$



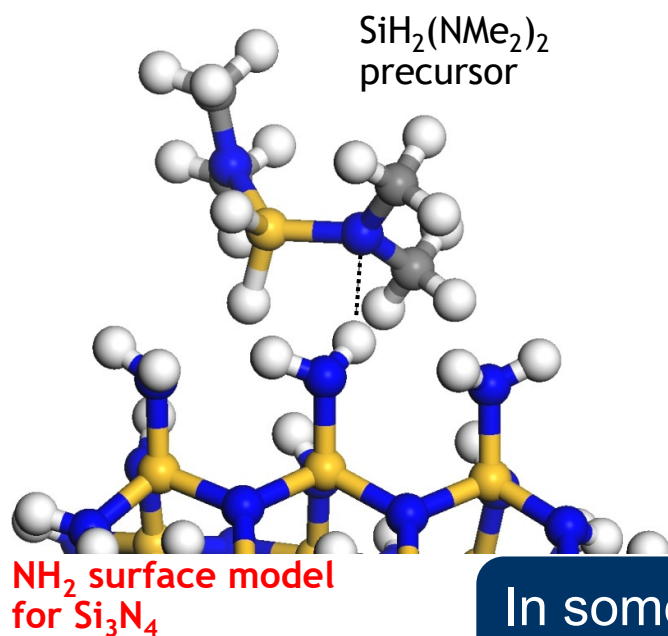
Major difference is in strength of initial molecular adsorption, not in choice of precursor

C. M. Murray *et al.*, *ACS Appl. Mater. Interf.* **6**, 10534 (2014).

[www.tyndall.ie](http://www.tyndall.ie)



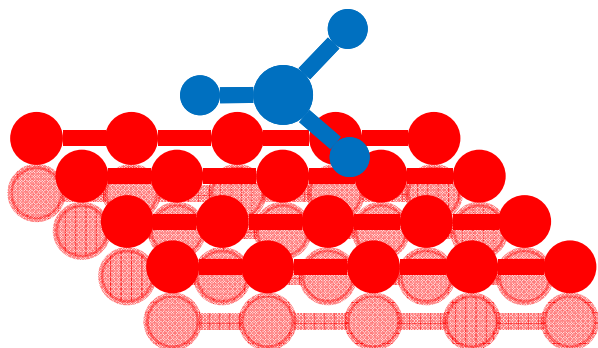
## Reactivity of ALD $\text{SiO}_2$ versus $\text{Si}_3\text{N}_4$



In some cases, simple model of isolated adsorbate on surface is adequate to explain growth behaviour.



# Accuracy of DFT results



Y. Maimaiti

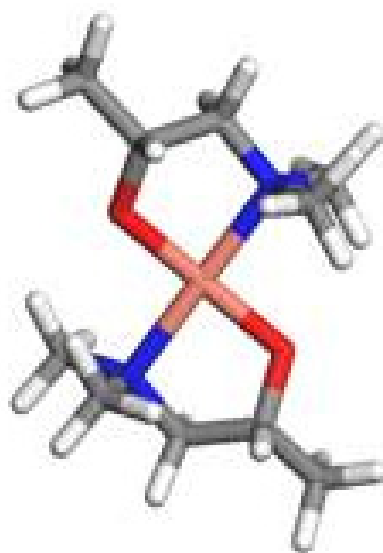
**Importance of van der Waals  
interactions during precursor  
adsorption**

**ALD of Cu**



# vdW interactions during adsorption

How do Cu precursors adsorb onto the growing Cu surface?



**Cu-O** break?

**Cu-N** break?

→ form **Cu-Cu**

**Cu-O** break?

**Cu-N** break?

...and **vdW** interactions?

**Cu(dmap)<sub>2</sub>**

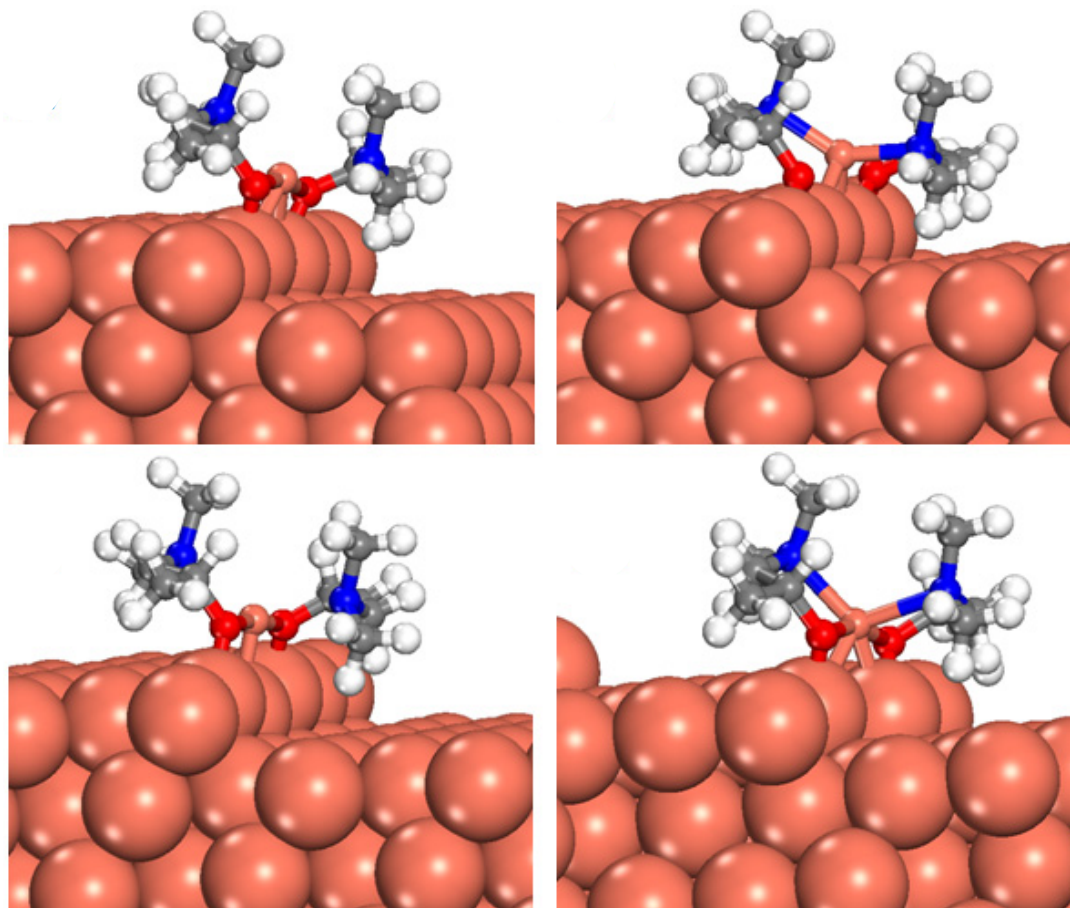
dmap=dimethyl-amino-propoxide

Cu=pink, O=red, N=blue, C=grey, H=white



## vdW interactions during adsorption

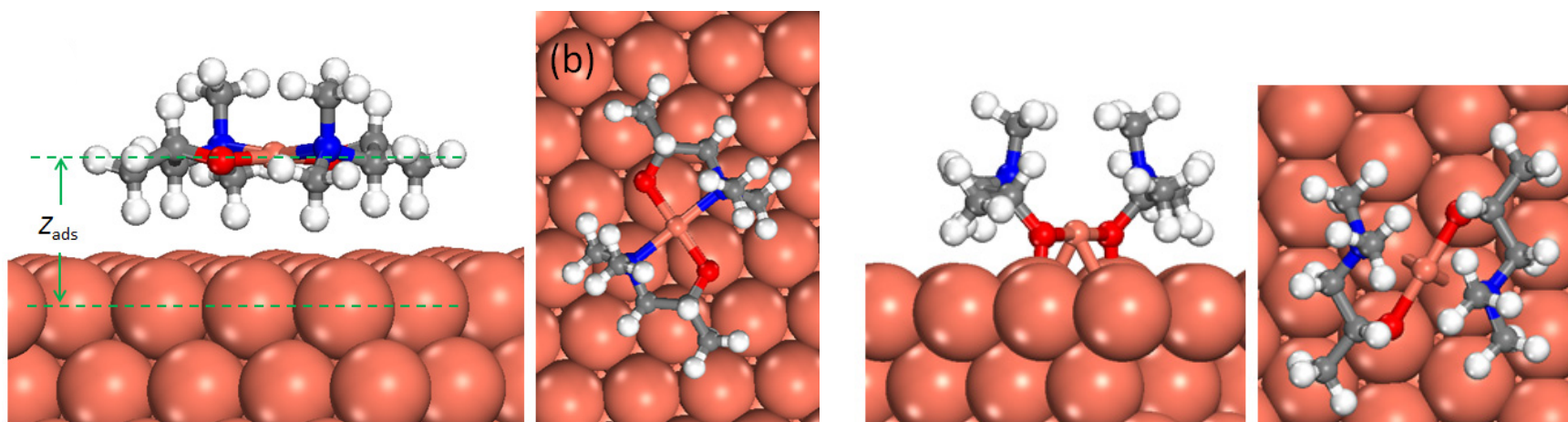
Chemisorption onto edge or kink for all functionals, regardless of vdW treatment:





## vdW interactions during adsorption

Adsorption onto flat Cu(111) is sensitive to van der Waals forces, alignment of Cu-O and cleavage of Cu-N:



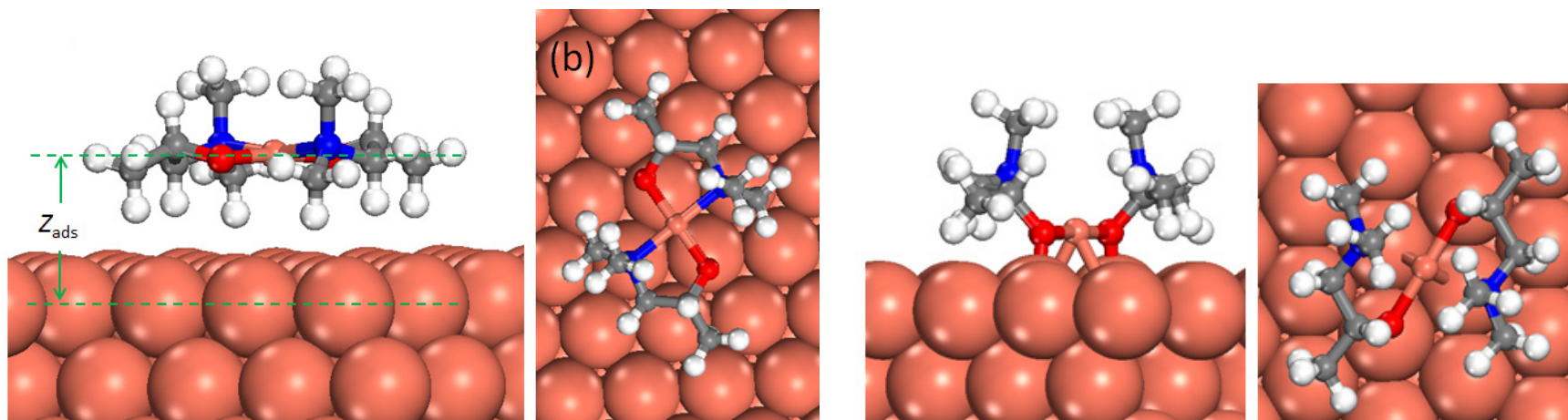
	$\Delta E(\text{physisorbed})$	$\Delta E(\text{chemisorbed})$
PBE	-0.4 eV (3 sites)	-1.5 eV (1 site)
vdW-DF2	-1.0 eV (3 sites)	-2.0 eV (1 site)
optB88-vdW	-1.6 eV (1 site)	-3.2 eV (3 sites)
PBE-D3	no sites	-3.5 eV (4 sites)





## vdW interactions during adsorption

Adsorption onto flat Cu(111) is sensitive to van der Waals forces, alignment of Cu-O and cleavage of Cu-N:

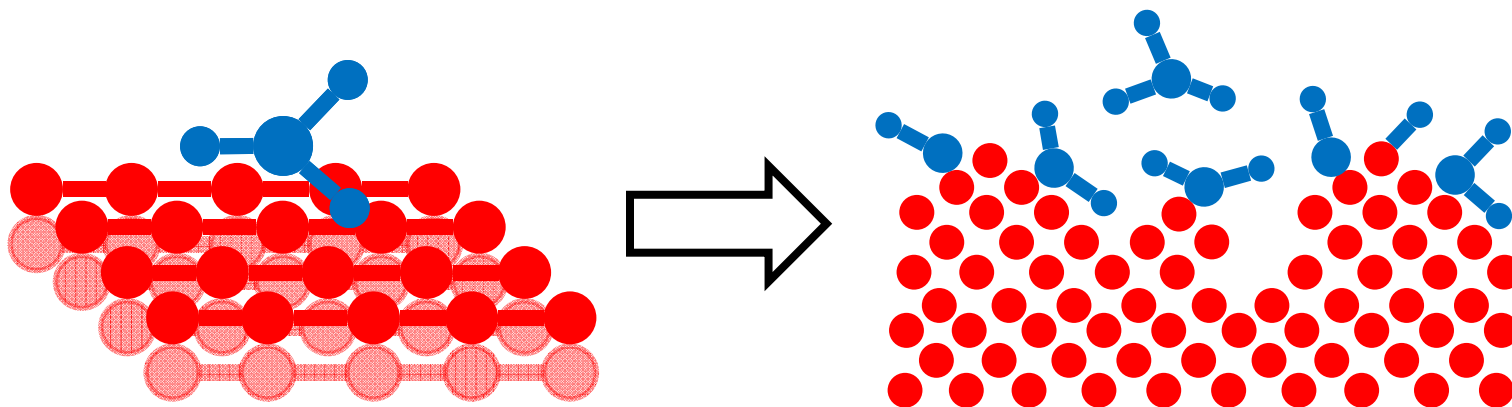


No consensus yet on how best to incorporate vdW contribution to adsorption.





# Accuracy of DFT results



Isolated ligands can persist on surface for entire cycle



M. Shirazi

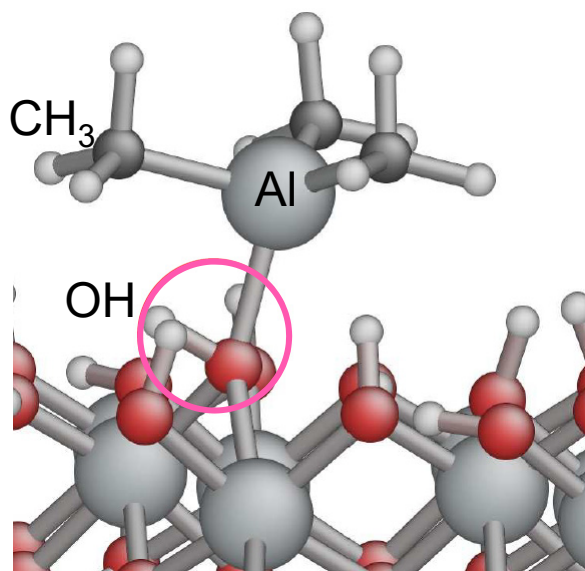
ALD of  $\text{Al}_2\text{O}_3$



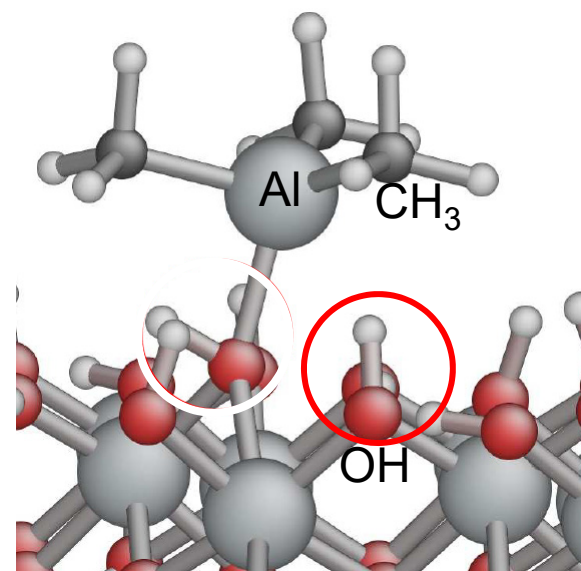


## Cooperative effect in surface kinetics

**TMA pulse:** Computed activation energies for transfer of  $H^+$  from surface-OH to adsorbate- $CH_3$ , resulting in desorption of  $CH_4$ :



$E_a = 0.28 \text{ eV}$   
from 3-coordinate OH

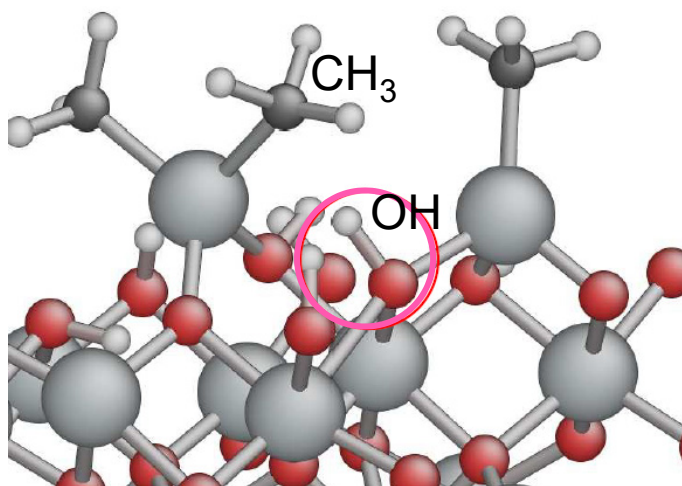


$E_a = 0.74 \text{ eV}$   
from 2-coordinate OH

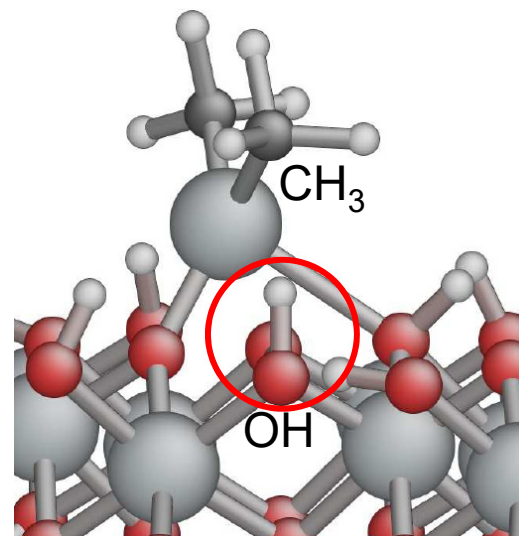


## Cooperative effect in surface kinetics

**TMA pulse:** Computed activation energies for transfer of  $H^+$  from surface-OH to adsorbate- $CH_3$ , resulting in desorption of  $CH_4$ :



$E_a = 0.26$  eV  
from 3-coordinate OH

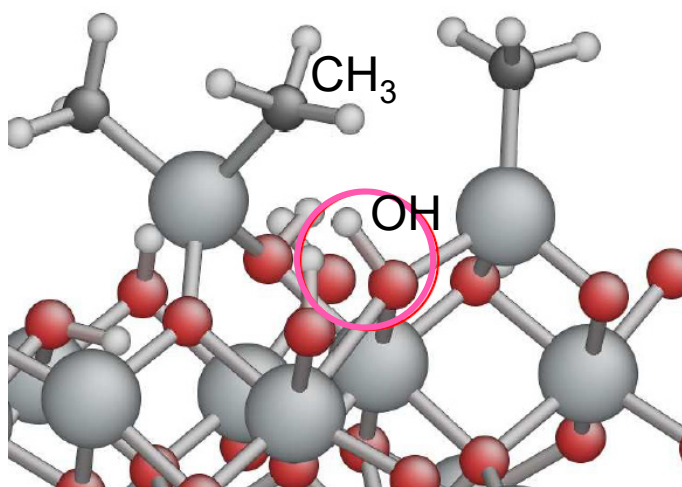


**no reaction**  
from 2-coordinate OH

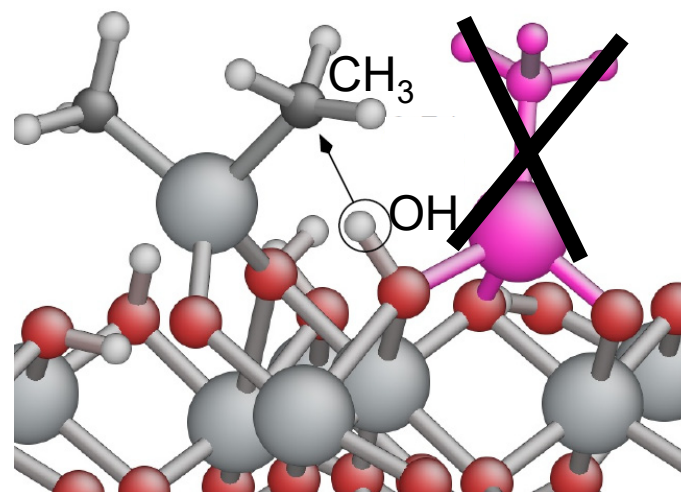


## Cooperative effect in surface kinetics

**TMA pulse:** Computed activation energies for transfer of  $\text{H}^+$  from surface-OH to adsorbate- $\text{CH}_3$ , resulting in desorption of  $\text{CH}_4$ :



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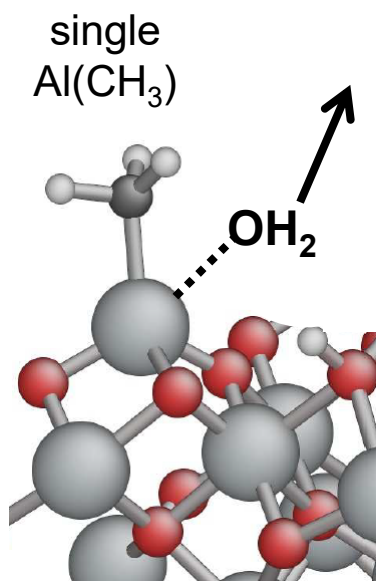
$E_a = 0.74 \text{ eV}$   
from 2-coordinate OH  
when neighbouring  $\text{AlCH}_3$  is missing

**Cooperative effect:** neighbouring adsorbate increases coordination number of O and thus increases Brønsted acidity of OH.



# Cooperative effect in surface kinetics

H<sub>2</sub>O pulse



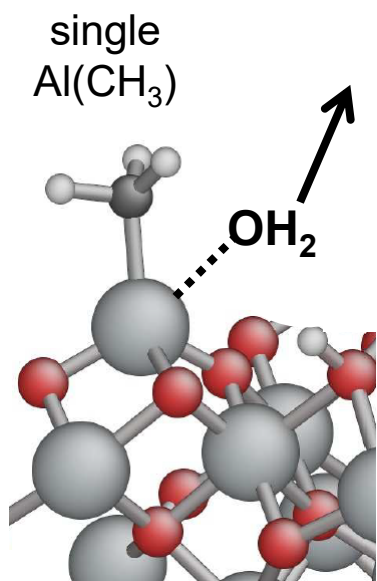
**No chemisorption**  
of single H<sub>2</sub>O  
to single Al(CH<sub>3</sub>)

large grey=Al, red=O, white=H, small grey=C

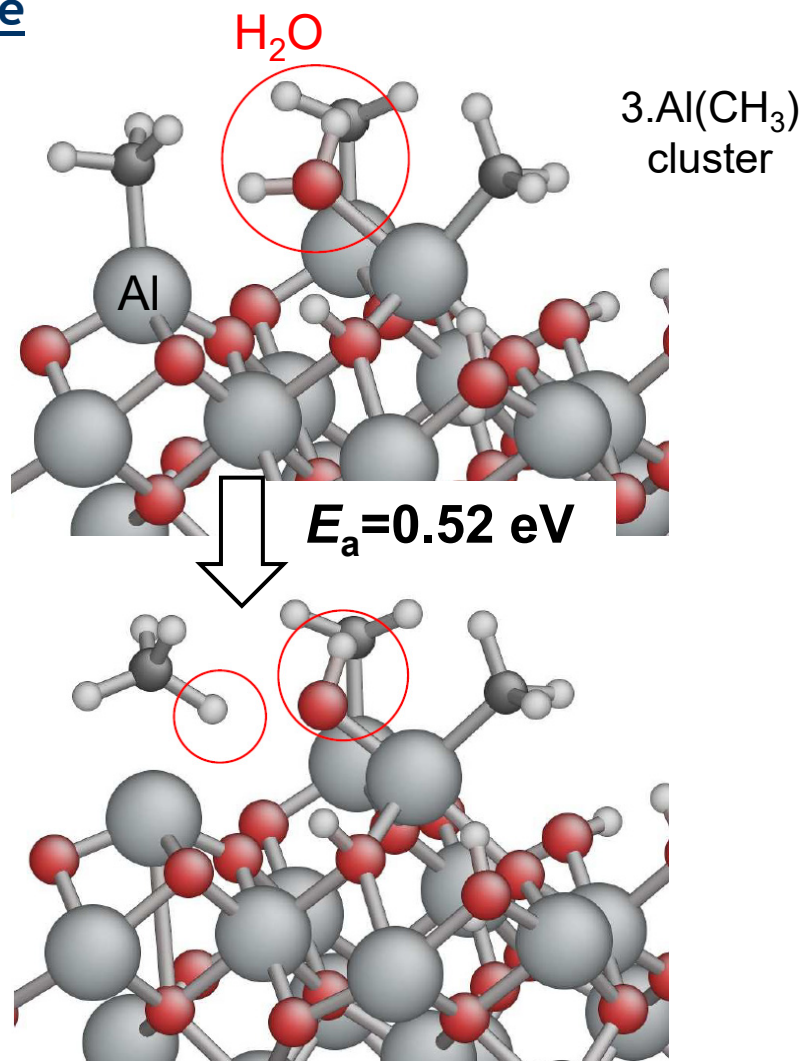


# Cooperative effect in surface kinetics

H<sub>2</sub>O pulse



**No chemisorption**  
of single H<sub>2</sub>O  
to single Al(CH<sub>3</sub>)

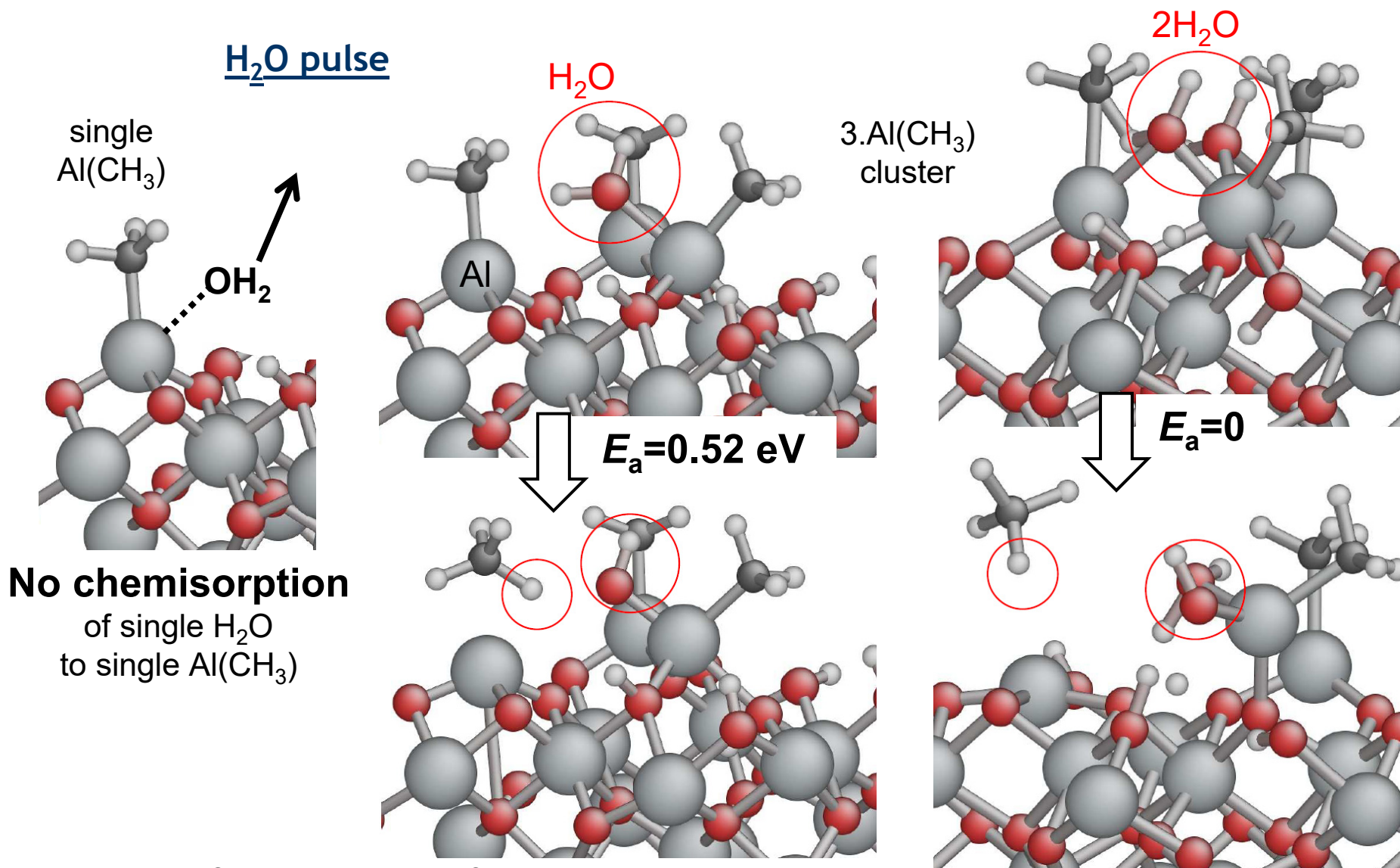


large grey=Al, red=O, white=H, small grey=C





# Cooperative effect in surface kinetics

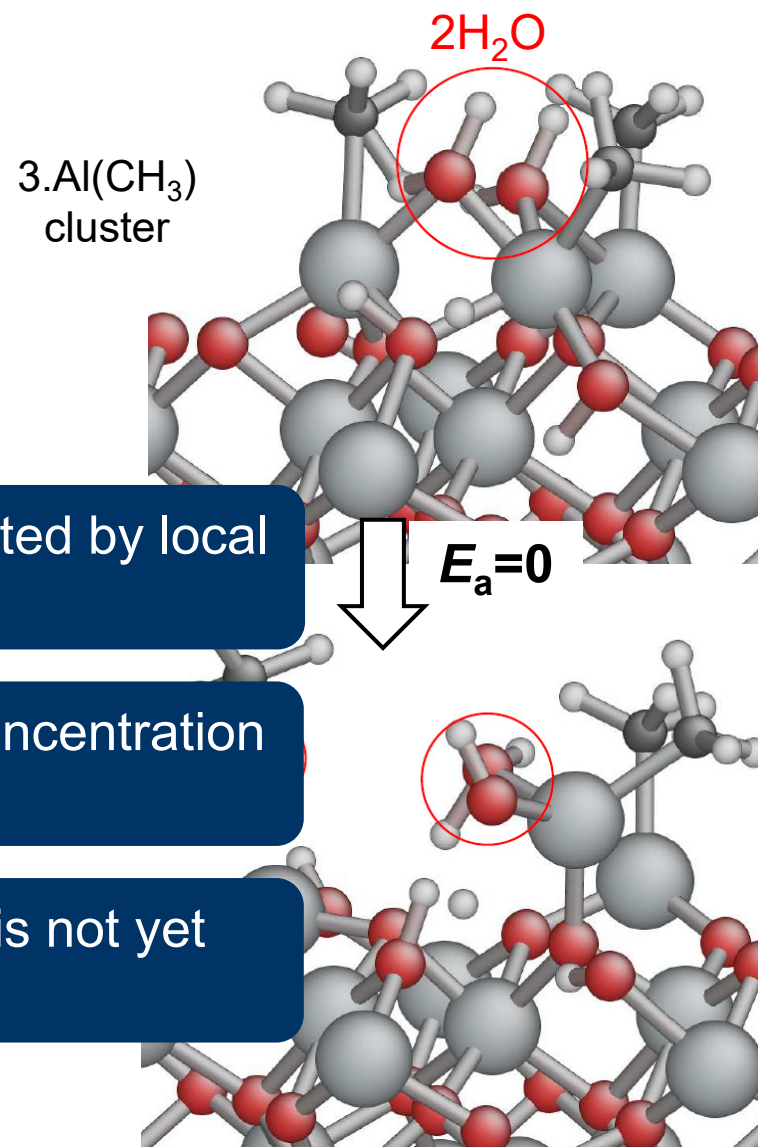


large grey=Al, red=O, white=H, small grey=C





## Cooperative effect in surface kinetics



Kinetics of surface reactions strongly affected by local environment.

Reactions are accelerated by high local concentration of ligands.

Effect of local film morphology on kinetics is not yet known.

large grey=Al, red=O, white=H, small grey=C

# What is limiting low-temperature atomic layer deposition of $\text{Al}_2\text{O}_3$ ? A vibrational sum-frequency generation study

 V. Vandalon<sup>a)</sup> and W. M. M. Kessels<sup>a)</sup>

Appl. Phys. Lett. **108**, 011607 (2016); doi: <http://dx.doi.org/10.1063/1.4939654>

PDF

ABSTRACT

FULL TEXT

FIGURES

SUPPLEM

Atomic layer deposition · Chemical reaction cross sections · Hydrogen reactions

## ABSTRACT

The surface reactions during atomic layer deposition of  $\text{Al}(\text{CH}_3)_3$  and  $\text{H}_2\text{O}$  have been studied with vibrational sum-frequency generation spectroscopy to reveal what is limiting the growth at

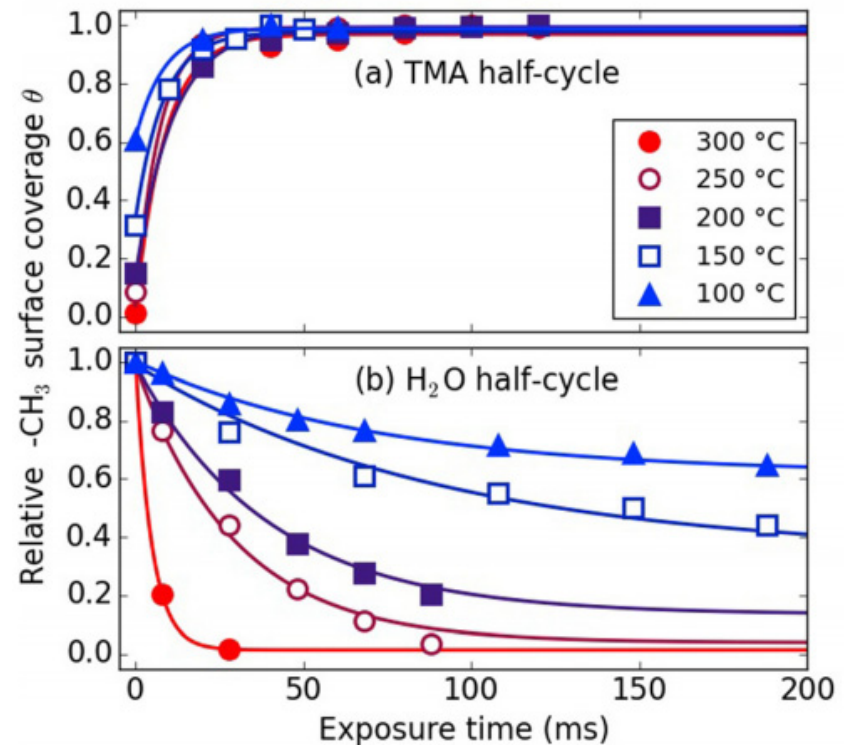
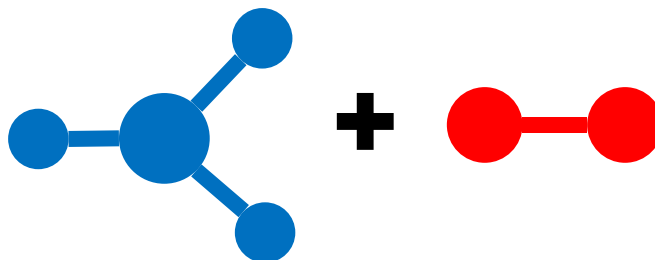


FIG. 3. Relative  $-\text{CH}_3$  surface coverage  $\theta$ , extracted from the BB-SFG spectra as a function of TMA and  $\text{H}_2\text{O}$  exposure for various temperatures. The solid lines represent fits to the data in order to extract information about the reaction kinetics.



# What theory can tell us about ALD mechanism



## SUCCESSSES:

Screening based on intrinsic chemistry;  
Oligomer formation during MLD

## CHALLENGES:

**Volatility?**

**Thermal stability?**

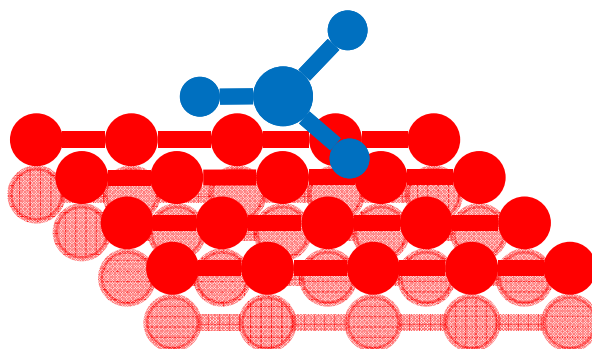


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# What theory can tell us about ALD mechanism



## SUCCESSSES:

Sketch out reaction mechanism;  
Account for many experimental anomalies

## CHALLENGES:

Accuracy of weak interactions;  
Need more realistic surface geometries for kinetics

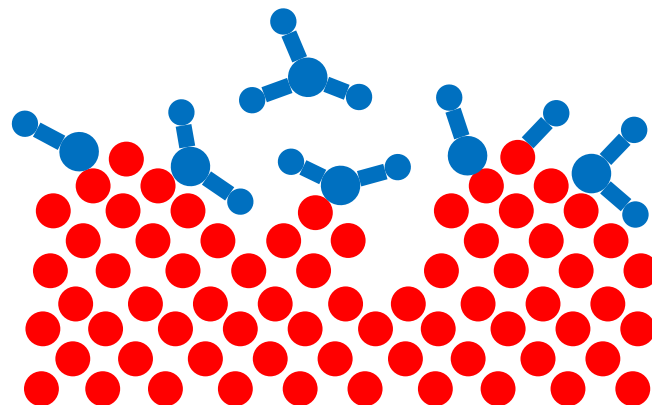


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# What theory can tell us about ALD mechanism



## SUCCESSSES:

More accurate activation energies;  
Indications of how morphology evolves

## CHALLENGES:

**Automatic search?**



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