# Structural Biology Methods

Fall 2024

Lecture #3

#### Scattering by a crystal

Suppose that the crystal has translation vectors  $\mathbf{a}$ ,  $\mathbf{b}$ , and  $\mathbf{c}$  and contains a large number of unit cells:  $n_1$  in the  $\mathbf{a}$  direction,  $n_2$  in the  $\mathbf{b}$  direction, and  $n_3$  in the  $\mathbf{c}$ 

direction (Figure 4.14). a t.a+u.b+v.cThe scattering of this unit cell with O as origin is:  $F(S)\exp[2\pi i t.a.S]\exp[2\pi i u.b.S]\exp[2\pi i v.c.S]$ 

The scattering of this unit cell

with O as origin is F(S)

Figure 4.14. A crystal contains a large number of identical unit cells. Only two of them are drawn in this figure.

To obtain the scattering by the crystal, we must add the scattering by all unit cells with respect to a single origin. We choose the origin O in Figure 4.14. For a unit cell with its own origin at position  $t \cdot \mathbf{a} + u \cdot \mathbf{b} + v \cdot \mathbf{c}$ , in which t, u, and v are whole numbers, the scattering is

$$\mathbf{F}(\mathbf{S}) \times \exp[2\pi i t \mathbf{a} \cdot \mathbf{S}] \times \exp[2\pi i u \mathbf{b} \cdot \mathbf{S}] \times \exp[2\pi i v \mathbf{c} \cdot \mathbf{S}].$$

The total wave K(S) scattered by the crystal is obtained by a summation over all unit cells:

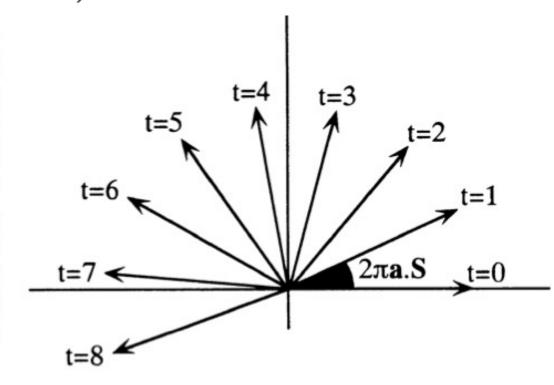
$$\mathbf{K}(\mathbf{S}) = \mathbf{F}(\mathbf{S}) \times \sum_{t=0}^{n_1} \exp[2\pi i t \mathbf{a} \cdot \mathbf{S}] \times \sum_{u=0}^{n_2} \exp[2\pi i u \mathbf{b} \cdot \mathbf{S}] \times \sum_{v=0}^{n_3} \exp[2\pi i v \mathbf{c} \cdot \mathbf{S}].$$

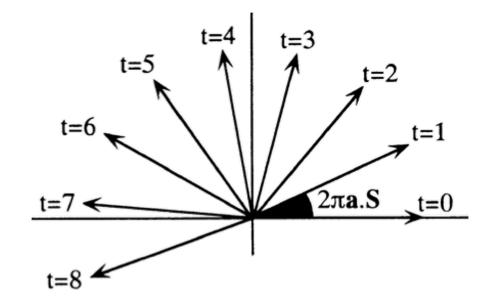
The total wave K(S) scattered by the crystal is obtained by a summation over all unit cells:

$$\mathbf{K}(\mathbf{S}) = \mathbf{F}(\mathbf{S}) \times \sum_{t=0}^{n_1} \exp[2\pi i t \mathbf{a} \cdot \mathbf{S}] \times \sum_{u=0}^{n_2} \exp[2\pi i u \mathbf{b} \cdot \mathbf{S}] \times \sum_{v=0}^{n_3} \exp[2\pi i v \mathbf{c} \cdot \mathbf{S}].$$

Because  $n_1$ ,  $n_2$ , and  $n_3$  are very large, the summation  $\sum_{t=0}^{n_1} \exp[2\pi i t \mathbf{a} \cdot \mathbf{S}]$  and the other two over u and v are almost always equal to zero unless  $\mathbf{a} \cdot \mathbf{S}$  is an integer h,  $\mathbf{b} \cdot \mathbf{S}$  is an integer k, and  $\mathbf{c} \cdot \mathbf{S}$  is an integer l. This is easy to understand if we regard  $\exp[2\pi i t \mathbf{a} \cdot \mathbf{S}]$  as a vector in the Argand diagram with a length of 1 and a phase angle  $2\pi t \mathbf{a} \cdot \mathbf{S}$  (see Figure 4.15).

Figure 4.15. Each arrow represents the scattering by one unit cell in the crystal. Because of the huge number of unit cells and because their scattering vectors are pointing in different directions, the scattering by a crystal is, in general, zero. However, in the special case that  $\mathbf{a} \cdot \mathbf{S}$  is an integer h, all vectors point to the right and the scattering by the crystal can be of appreciable intensity.





Conclusion: A crystal does not scatter X-rays, unless

$$\mathbf{a} \cdot \mathbf{S} = h,$$
  
 $\mathbf{b} \cdot \mathbf{S} = k,$   
 $\mathbf{c} \cdot \mathbf{S} = l.$  (4.4)

These are known as the Laue conditions. h, k, and l are whole numbers, either positive, negative, or zero. The amplitude of the total scattered wave is proportional to the amplitude of the structure factor F(S) and the number of unit cells in the crystal.

#### **Calculation of electron density**

The structure factor is a function of the electron density distribution in the unit cell:

$$\mathbf{F}(\mathbf{S}) = \sum_{j} f_{j} \exp[2\pi i \mathbf{r}_{j} \cdot \mathbf{S}]. \tag{4.3}$$

$$\mathbf{F}(\mathbf{S}) = \int_{\text{cell}} \rho(\mathbf{r}) \exp[2\pi i \mathbf{r}_j \cdot \mathbf{S}] d\nu. \tag{4.8}$$

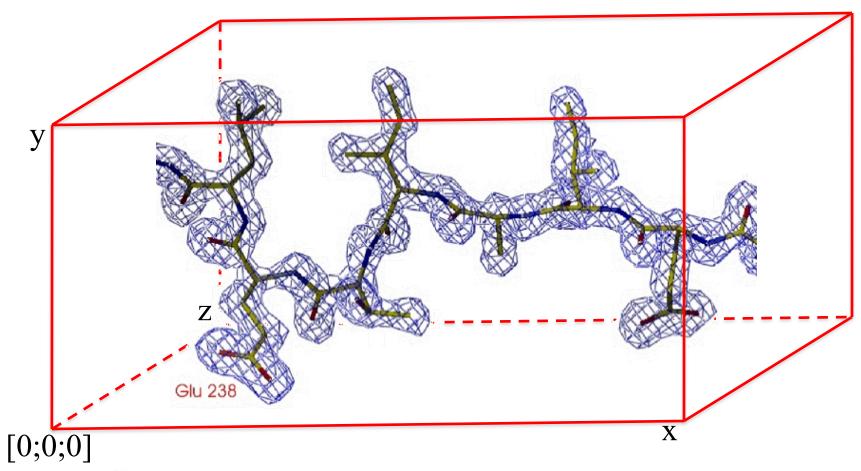
where  $\rho(\mathbf{r})$  is the electron density at position  $\mathbf{r}$  in the unit cell. If x, y, and z are fractional coordinates in the unit cell ( $0 \le x < 1$ ; the same for y and z) and V is the volume of the unit cell, we have

$$dv = V \cdot dx \, dy \, dz$$

and

$$\mathbf{r} \cdot \mathbf{S} = (\mathbf{a} \cdot x + \mathbf{b} \cdot y + \mathbf{c} \cdot z) \cdot \mathbf{S} = \mathbf{a} \cdot \mathbf{S} \cdot x + \mathbf{b} \cdot \mathbf{S} \cdot y + \mathbf{c} \cdot \mathbf{S} \cdot z$$
  
=  $hx + ky + lz$ .

# Information from X-ray diffraction experiment



$$\rho(x \ y \ z) = \frac{1}{V} \sum_{h} \sum_{k} \sum_{l} \left| F(h \ k \ l) \right| \exp\left[ -2\pi i (hx + ky + lz) + i\alpha(h \ k \ l) \right]$$

# **Temperature (B) factor**

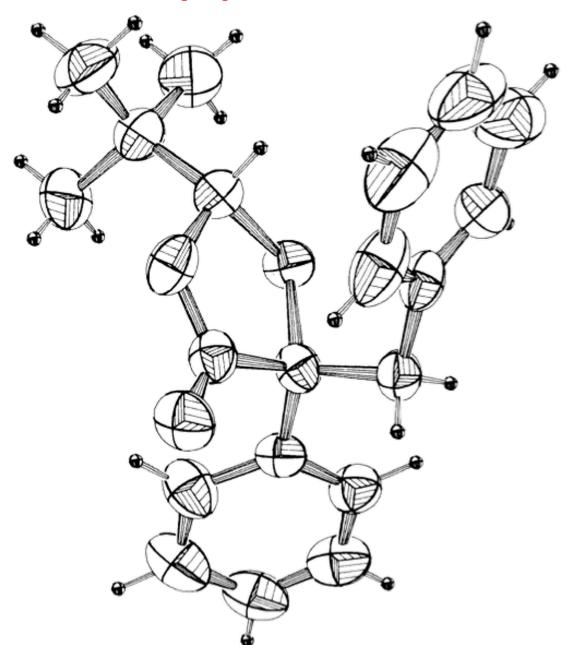


Figure 4.22. The plot of an organic molecule with 50% probability of thermal ellipsoids. (Reproduced with permission from Strijtveen and Kellogg © 1987 Pergamon Press PLC.)

$$F(h k l) = V \int_{x=0}^{1} \int_{y=0}^{1} \int_{z=0}^{1} \rho(x y z) \exp[2\pi i (hx + ky + lz)] dx dy dz.$$
 (4.9)

 $\mathbf{F}(h \, k \, l)$  is the Fourier transform of  $\rho(x \, y \, z)$ , but the reverse is also true:  $\rho(x \, y \, z)$  is the Fourier transform of  $\mathbf{F}(h \, k \, l)$  and, therefore,  $\rho(x \, y \, z)$  can be written as a function of all  $\mathbf{F}(h \, k \, l)$ :

$$\rho(x \ y \ z) = \frac{1}{V} \sum_{h} \sum_{k} \sum_{l} \mathbf{F}(h \ k \ l) \exp[-2\pi i (hx + ky + lz)]. \tag{4.10}$$

The Laue conditions tell us that diffraction occurs only in discrete directions and, therefore, in Equation (4.10), the integration has been replaced by a summation. Because  $\mathbf{F} = [F] \exp[ia]$ , we can also write

$$\rho(x \ y \ z) = \frac{1}{V} \sum_{h} \sum_{k} \sum_{l} \left| F(h \ k \ l) \right| \exp\left[ -2\pi i \frac{(hx + ky + lz)}{(hx + ky + lz)} + i \frac{\alpha(h \ k \ l)}{(4.11)} \right]$$

#### Notes

- 1.  $\mathbf{F}(h \, k \, l)$  is the Fourier transform of the electron density  $\rho(x \, y \, z)$  in the entire unit cell. Often the unit cell contains more than one molecule. Then  $\mathbf{F}(h \, k \, l)$  is composed of the sum of the transforms of the separate molecules at position  $(h \, k \, l)$  in reciprocal space.
- 2. Because of the crystallographic repeat of the unit cells, the value of the transform  $\mathbf{F}(h \ k \ l)$  is zero in between the reciprocal space positions  $(h \ k \ l)$ . If there were no crystallographic repeat, the transform would be spread over the entire reciprocal space and its value is not restricted to reciprocal space positions  $(h, k \ l)$ .

### Intensity diffracted by a crystal

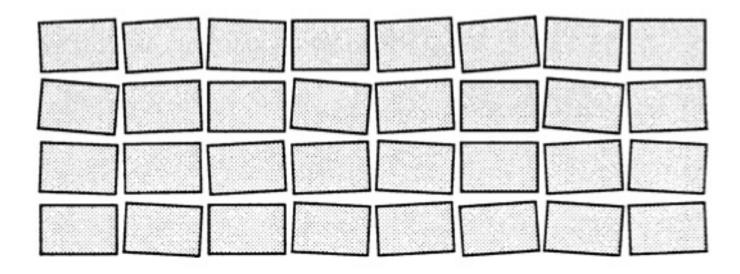


Figure 4.29. Most crystals are imperfect and can be regarded as being composed of small mosaic blocks.

We have the following assumptions:

- Apart from ordinary absorption, the intensity I<sub>0</sub> of the incident beam is the same throughout the crystal.
- The mosaic blocks are so small that a scattered wave is not scattered again (i.e., multiple scattering does not occur).
- 3. The mosaic blocks scatter independently of each other.

With these assumptions, the expression for I (int., h k l), if the crystal is rotated with an angular velocity  $\omega$  through the reflection position, is

$$I(\text{int., } h \, k \, l) = \frac{\lambda^3}{\omega \cdot V^2} \times \left(\frac{e^2}{mc^2}\right)^2 \times V_{\text{cr}} \times I_0 \times L \times P \times T_{\text{r}} \times |F(h \, k \, l)|^2.$$

$$(4.32)$$

 $\lambda$  – wavelength

 $\omega$  – angular velocity of crystal

rotation

V – unit cell volume

e – electron charge

m – electron mass

c – speed of light

V<sub>cr</sub> – crystal volume

I<sub>0</sub> – intensity of the excitation beam

L – Lorentz coefficient

P – polarization coefficient

T<sub>r</sub> – transmission coefficient

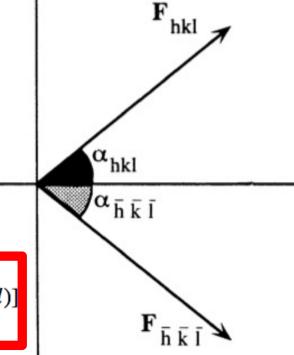
|F(hkl)| - structure factor amplitude

#### Friedel pairs

$$\mathbf{F}(h \, k \, l) = V \int_{\text{cell}} \rho(x \, y \, z) \exp[2\pi i (hx + ky + lz)] \, dx \, dy \, dz$$

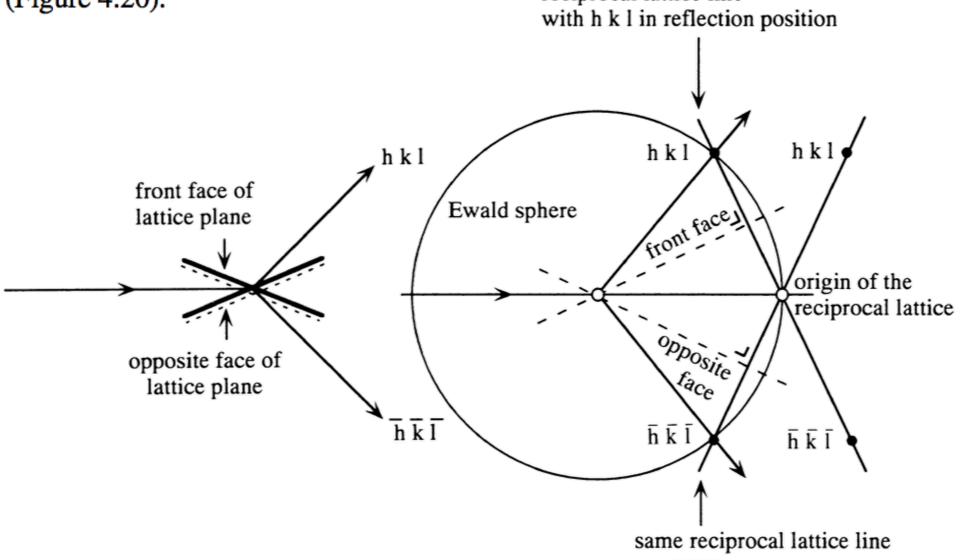
$$\mathbf{F}(\bar{h}\,\bar{k}\,\bar{l}) = V \int_{\text{cell}} \rho(x\,y\,z) \exp[2\pi i(-hx - ky - lz)] \,dx \,dy \,dz. \tag{4.25}$$

Figure 4.24. Argand diagram for the structure factors of the reflections  $\mathbf{F}(h \ k \ l)$  and  $\mathbf{F}(\bar{h} \ \bar{k} \ \bar{l})$ .



$$\rho(x \ y \ z) = \frac{2}{V} \sum_{hkl=0}^{+\infty} |F(h \ k \ l)| \cos[2\pi(hx + ky + lz) - \alpha(h \ k \ l)]$$

One more comment on lattice planes: If the beam  $h \, k \, l$  corresponds to reflection against one face (let us say the front) of a lattice plane, then  $(\bar{h}\bar{k}\,\bar{l})$  [or (-h,-k,-l) corresponds to the reflection against the opposite face (the back) of the plane (Figure 4.20).



with h k l in reflection position

### Symmetry in the diffraction pattern

#### 4.12.1. A 2-Fold Axis Along *y*

If a 2-fold axis through the origin and along y is present, then the electron density  $\rho(x \ y \ z) = \rho(\bar{x} \ y \ \bar{z})$  (Figure 4.25). Therefore,

$$\mathbf{F}(h \, k \, l) = V \int_{\substack{\text{asymm} \\ \text{unit}}} \rho(x \, y \, z) \{ \exp[2\pi i (hx + ky + lz)] + \exp[2\pi i (-hx + ky - lz)] \} \, dx \, dy \, dz$$

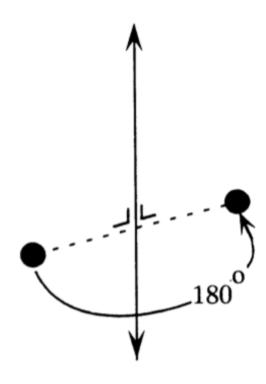
$$(4.26)$$

The integration in Eq. (4.26) is over one asymmetric unit (half of the cell), because the presence of the second term under the integral takes care of the other half of the cell.

$$\mathbf{F}(\bar{h}\,k\,\bar{l}) = V \int_{\substack{\text{asymm} \\ \text{unit}}} \rho(x\,y\,z) \{\exp[2\pi i(-hx + ky - lz)] + \exp[2\pi i(hx + ky + lz)]\} \,dx\,dy\,dz$$

$$(4.27)$$

It follows that  $\mathbf{F}(h \, k \, l) = \mathbf{F}(\bar{h} \, k \, \bar{l})$  and also  $I(h \, k \, l) = I(\bar{h} \, k \, \bar{l})$ ,



#### 4.12.2. A 2-Fold Screw Axis Along y

For a 2-fold screw axis along y (Figure 4.26),

$$\rho(x \ y \ z) = \rho\{\bar{x}(y+1/2)\bar{z}\}$$

$$\text{term I} \downarrow$$

$$\mathbf{F}(h \ k \ l) = V \int_{\substack{\text{asymm} \\ \text{unit}}} \rho(x \ y \ z) \{\exp[2\pi i (hx + ky + lz)]$$

$$+ \exp[2\pi i (-hx + k(y+1/2) - lz)]\} \ dx \ dy \ dz \qquad (4.28)$$

$$\text{term III} \downarrow$$

$$\mathbf{F}(\bar{h} \ k \ \bar{l}) = V \int_{\substack{\text{asymm} \\ \text{unit}}} \rho(x \ y \ z) \{\exp[2\pi i (-hx + ky - lz)]$$

$$+ \exp[2\pi i (hx + k(y+1/2) + lz)]\} \ dx \ dy \ dz. \qquad (4.29)$$

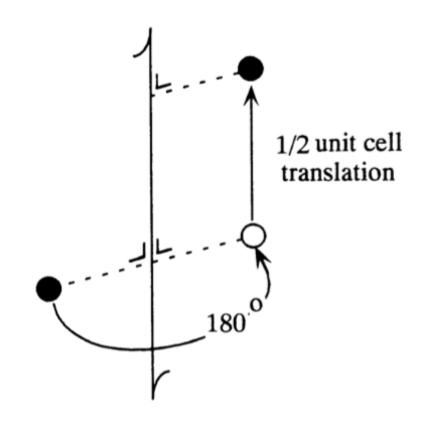
$$\text{term IV} \uparrow$$

In Equation (4.28), term II is

$$\exp\{2\pi i[-hx + k(y+1/2) - lz]\} = \exp[2\pi i(-hx + ky - lz + 1/2k)].$$

For k even, this is equal to term III in Equation (4.29). The same is true for term IV in Equation (4.29) and term I in Equation (4.28). Therefore, when k is even,  $\mathbf{F}(h k l) = \mathbf{F}(\bar{h} k \bar{l})$  and  $I(h k l = I(\bar{h} k \bar{l}))$ . When k is odd, terms I and IV have a difference of  $\pi$  in their phase angles:  $2\pi(hx + ky + lz)$  and  $2\pi(hx + ky + lz + 1/2k)$ .

For F(hkl) with k odd:  $\mathbf{F}(\bar{\mathbf{h}}\; \mathbf{k}\; \bar{\mathbf{l}})$ 



## Systematic absences in P2(1)

$$\mathbf{F}(0 \, k \, 0) = V \int_{\text{asymm unit}} \rho(x \, y \, z) \{ \exp[2\pi i k y] + \exp[2\pi i k (y + 1/2)] \} \, dx \, dy \, dz.$$
(4.30)

When k is even, this is  $2 \times V \int \rho(x y z) \exp[2\pi i k y] dx dy dz$ . However, when k is odd, the two terms in Equation (4.30) cancel and  $\mathbf{F}(0 k 0) = 0$ 

